

# Fiber Bragg grating regeneration at 450 °C for improved high temperature sensing

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**Type-I fiber Bragg gratings photo-inscribed in hydrogen-loaded B/Ge co-doped silica singlemode optical fibers have been regenerated efficiently at 450 °C, which is the lowest temperature reported so far. The mechanical strength of the annealed fiber is preserved while ensuring temperature sensing of the regenerated gratings up to 900 °C. Unlike low temperature cycles ( $\leq 600$  °C), an annealing process at higher temperatures revealed faster regeneration for strong gratings. Changes in grating strength were also measured before regeneration cycle. These behaviors suggest the contribution of different mechanisms to the regeneration process with different relative dynamics. © 2019 Optical Society of America**

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Fiber Bragg gratings (FBGs) have become a standard for sensing purposes [1]. They even outperform their electric counterparts in many application fields. In harsh environment characterized by extreme physical and/or chemical conditions like in the automotive sector (combustion engine vehicles), electric power generation (gas turbines) or oil and nuclear industries, a precise control of high temperature (over 400 °C) and/or strain is required [2, 3]. Unfortunately, under these severe conditions, FBGs produced by standard UV inscription techniques suffer from practical limitations. Indeed, the reflectivity of type-I FBG starts to decay when temperature reaches 400 °C [4]. To overcome this limitation, several techniques have been developed to increase the resistance of FBGs to high temperature. Grating types proposed so far are Type IIa [5], Sn-doped silica gratings [6], chemical composition gratings (CCG) [7], gratings produced by IR femtosecond laser pulses [3, 8] and regenerated gratings (RFBG) in gas loaded single-mode fiber and in loaded and unloaded photosensitive fibers [9-14]. This last technique consists in applying to an initial FBG (seed) a heating cycle that will erase it first and rebuild a new one on its 'footsteps'. Despite all these

efforts, the sensing properties of FBGs at high temperatures as well as their long-term stability remains a controversial subject that requires further investigation. Glass-softening [9, 12-14] is believed to be the key limitation for FBGs technology in this field of application. The very high temperature (700-1100 °C) applied during the annealing process reduces the mechanical sustainability of the fiber. Ceramic and metal tubes applied before the annealing process have been demonstrated as a possible protection [15]. Most of the proposed solutions in the literature are oriented towards the improvement of temperature sensing capabilities accounting for regeneration efficiency (ratio between regenerated and seed grating reflectivities) and/or long-term thermal stability. Recently, a pre-annealing at 700 °C was demonstrated to generate a high refractive index modulation of the regenerated grating in standard single-mode fiber [16]. An isothermal annealing was also investigated to lower the regeneration temperature and increase the long-term stability of the gratings [17]. A regeneration temperature of 680 °C and good efficiency were demonstrated in hydrogen loaded Ge/B co-doped fiber [18]. Only few studies were dedicated to the strain sensing performances of the RFBGs [2].

In this work, we investigate two annealing approaches. In the first one [18], temperature is increased monotonously until a plateau (700 – 1100 °C). During this process the gratings go through 'fast' regeneration cycle where the so-called regeneration temperature ( $T_{Reg}$ ), corresponding to the total erasure of the FBGs, is determined. In the second method, we set the plateau at a temperature well below  $T_{Reg}$  where FBGs undergo 'slower' regeneration cycle. The main goal is to lower as much as possible the temperature of the regeneration process within acceptable time period so as to preserve the mechanical robustness of the FBGs. Performance parameters like the regeneration efficiency, strain and temperature sensitivities of the RFBGs as well as its high temperature operation limit were determined.

Prior to the inscription process of FBGs the photosensitive optical fiber (PS1250/1500) including 10 mol % of GeO<sub>2</sub> and 14–18 mol % of B<sub>2</sub>O<sub>3</sub> is hydrogen-loaded under ~200 bar and 60 °C

for 30 hours. Type-I FBGs (seed) were then produced with phase mask technique and 7 ns pulse duration ArF Excimer laser emitting at 193 nm with 5 mJ energy and 50 Hz repetition rate. Different FBGs of 10 mm long with varying transmission minimum ( $T_{min}$ ) at Bragg wavelength ( $\lambda_B$ ) were produced (2.0 - 23.5 dB). For higher efficiency, regeneration cycles were conducted without any pre-annealing as reported in [18]. FBGs were placed in a tubular furnace able to reach 1200 °C with different heating rates. Both reflection and transmission spectra were monitored during and after the annealing process thanks to a FBG interrogator from *FiberSensing* of optimal acquisition (1 Hz). From the measured  $T_{min}$  we determined the reflectivity ( $R$ ). The grating strength, which is proportional to the integrated coupling constant  $ICC$  [4] is derived:

$$ICC = \tanh^{-1} \sqrt{R} = \tanh \sqrt{1 - T_{min}} \quad (1)$$

For comparison, the ICC normalized to its initial value ( $\eta_{Tmin}$ ) is used. When only the reflection spectrum is recorded, the grating strength can be estimated from the full width at half maximum FWHM ( $\Delta\lambda$ ) of the grating normalized to its initial value ( $\Delta\lambda_0$ ) [19]:

$$\eta_{FWHM} = \frac{\Delta n_m}{\Delta n_{m0}} \equiv \frac{\Delta\lambda}{\Delta\lambda_0} \quad (2)$$

where  $\Delta n_m$  and  $\Delta n_{m0}$  denote respectively the refractive index modulation of the grating and its initial value. For each FBG,  $\Delta n_{m0}$  is calculated using coupled mode theory and reported in Table 1.

Typical transmission and reflection spectra before (Seed FBG) and after regeneration (RFBG) are depicted in figure 1. The reflectivity is 65 %, and 14 % for the seed and regenerated FBG, respectively yielding a regeneration efficiency ( $\eta = R_{RFBG}/R_0$ ) of 21.5 %.

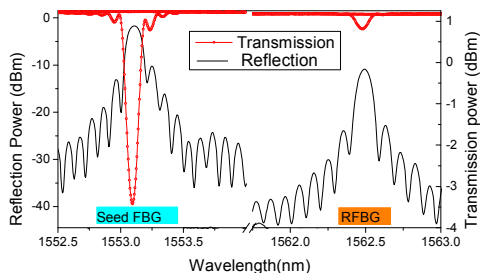


Fig. 1. Reflection and transmission powers of seed and RFBG with initial reflectivity  $R_0 = 65\%$ .

In Figure 2, we show the evolution versus time of temperature, normalized reflection power ( $\eta_{Rp}$  in blue) and grating strength calculated from Eq. 1 ( $\eta_{Tmin}$  in red) and Eq. 2 ( $\eta_{FWHM}$  in black). The inset in figure 2 is a close-up of the graph between 100 and 120 minutes (gray band). This range depicts the regeneration process where the FBG disappears and starts to rebuild. The area with sparse filling shows that R power ( $\eta_{Rp}$ ) can still be measured in contrary to  $T_{min}$  ( $\eta_{Tmin}=0$ ). Hence, to determine  $T_{Reg}$  it is better to use  $\eta_{Rp}$ . Before the FBG is erased, another important process occurs, the effect of which is well described by  $\eta_{Tmin}$  since the reflection power saturates for strong FBGs and the FWHM cannot provide with high precision the changes in the FBG strength. Hence, an exponential decay of  $\eta_{Tmin}$  is operated between 100 °C and 230 °C, then the decay slows down until 400 °C. The reflectivity of the FBG

is reduced by more than 40 %. Afterward, between 400 °C and 550 °C, the FBG recovers almost all of its initial strength before it stars to decay for the second time.

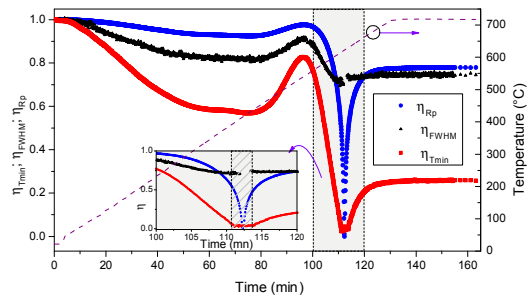


Fig. 2.  $\eta_{Tmin}$ ,  $\eta_{Rp}$  and  $\eta_{FWHM}$  versus time during the annealing cycle for FBG with  $R_0 = 65\%$ .

Similar low temperature changes (Fig. 2) have been reported in [20] and explained as a first regeneration regime, while in [21] FBGs instabilities are attributed to thermal annealing of boron-related refractive index in the fiber. Chisholm et al. proposed a model based on both theories [22]: thermal decay of the UV-induced refractive index change of the grating [4] and thermal annealing of the boron-related refractive index of the fiber [23]. The negative average change in refractive index is caused by the first mechanism starting at low temperature, while the second one operates a positive change in refractive index of the fiber core. This model did not take account of the regeneration mechanism since the maximum used experimental annealing temperature was 475 °C for 6 hours. In a recent work on the regeneration of long period gratings inscribed in boron-co-doped germanosilicate single-mode fiber [24], a nonlinear temperature response of the grating with three threshold points has been observed. Due to low transition temperature of germanium and boron, the observed effect has been correlated to the phase transition of glass ( $T_g$ ) in the core and inner cladding at  $\sim 500$  °C and  $\sim 250$  °C respectively, as well as the melting of inner cladding between 860 °C and 900 °C. In the light of all these reported results, a possible explanation of our experimental finding – as its modeling is beyond the scope of this paper – is that the first decrease in reflectivity is due to thermal decay of the UV-induced refractive index change [4-22]. The enhanced reflectivity around 500 °C can be associated to an increase in the core effective refractive index [22] close to the  $T_g$  of the fiber core (annealing of boron), which becomes much larger than the negative average change in refractive index due to decay of the UV-induced refractive index modulation. The observed second sharp decrease after 500 °C can be linked to a release of internal stress due to both UV inscription (periodic stress) and to high difference in doping concentration distribution between the fiber core and cladding [14]. This process takes the advantage over the boron annealing effect and brings the reflection and transmission power under the limit of detection. This turning situation over the boron annealing effect implies that at least two mechanisms are behind the UV-induced refractive index change: color center annealing (starting at low temperature) and stress-induced compaction [25]. Finally, the last increase in reflectivity, which corresponds to the rebuild of the FBGs, can be explained by thermal stabilization of internal and periodic stress relaxation and

crystallization ( $\alpha$ -quartz) that possibly occurred at low temperature because of stress induced-high pressure [14]. Therefore, in the first part of the experiments, we consider gratings of different reflectivities to determine their regeneration temperature ( $T_{Reg}$ ) or more precisely their annealing temperature threshold for fast regeneration process ( $\sim 10$  min) [18]. High temperature and therefore fast annealing cycle is considered. The FBGs are heated with a ramp of 5 °C/min up to 1100 °C.

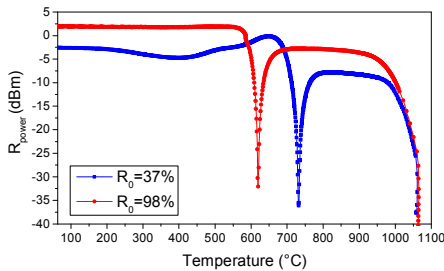


Fig. 3. Reflection power versus temperature for FBGs with  $R_0 = 37\%$  and  $R_0 = 98\%$ .

**Table 1.**  $R_0$ ,  $\Delta n_{m0}$ ,  $R_{Reg}$ , regeneration efficiency ( $\eta$ ) and  $T_{Reg}$ .

$R_0$ (%)	$\Delta n_{m0}$	$R_{Reg}$ (%)	$\eta$ (%)	$T_{Reg}$ (°C)
37	$4.7 \cdot 10^{-5}$	11	29.7	709
50	$5.8 \cdot 10^{-5}$	17	34.0	688
65	$7.4 \cdot 10^{-5}$	14	21.5	623
98	$1.78 \cdot 10^{-4}$	36	36.2	617

Figure 3 shows the  $R_{power}$  versus temperature for two FBGs with  $R_0$  equal to 37% and 98%. One can see that the FBGs undergo a regeneration cycle from 620 °C up to 800-900 °C, then they start to decay and erase completely at 1060 °C (corresponding to the softening of the fiber core). Moreover, the FBG with the highest  $R_0$  erases first and starts to regenerate as in [27]. This effect is outlined in table 1, where  $T_{Reg}$  and the reflectivity of the regenerated grating ( $R_{Reg}$ ) are summarized for seed FBGs of different  $R_0$  and corresponding  $\Delta n_{m0}$ . As explained before, annealing of internal and UV-induced periodic stresses are responsible for the FBG erasure after compensation of boron annealing effect. Highly reflective gratings because of higher UV exposure time, show higher stress effect. Therefore, at the activation energy of the process [25] the boron annealing effect is rapidly compensated in strong gratings, while it is still prevailing and screening the decrease in the weaker one.

In the second part of the experiments, we target to lower the annealing temperature to preserve the mechanical properties of the fiber and to compare the influence of the different proposed mechanisms involved in the fast regeneration cycle of FBGs. Two annealing temperatures below  $T_{Reg}$  are proposed: 600 °C and 450 °C. To reach the targeted temperatures, the FBG is heated rapidly with a rate of 12 °C/min. Once the temperature is reached, the process requires less than 1 hour for  $T=600$  °C, while 150

hours were needed for  $T=450$  °C. These regeneration temperatures are the lowest used so far in relatively short time period since 450 hours has been reported as a required time to decrease the regeneration temperature to 700 °C for FBGs in standard SMF28 [16]. As explained previously, the fiber composition, mainly boron doping, which decreases the transition temperature of the core [24] as well as 193 nm UV-induced defects type [13] contribute to lower the regeneration temperature. Figure 4 shows the annealing cycles at 600 °C and 450 °C for two FBGs with initial reflectivities 65% and 98%. The inset of the graph is a close-up showing the reflectivity changes, discussed previously, occurring when the temperature is still evolving linearly. Then the regeneration cycle occurs during the isothermal heating (at the targeted temperatures). It can also be observed, in contrast to the fast cycle, that FBG with  $R_0=65\%$  erases before the one with  $R_0=98\%$ . The Bragg wavelength  $\lambda_B$  was also measured at room temperature before and after regeneration cycles. All the investigated gratings (65%-98%) show a wavelength red shift of (1,002-0,670) nm and (0,500-0,200) nm for regeneration at 450 °C and 600 °C, respectively. This red shift is linked to boron annealing effect partially compensated by DC refractive index modulation decay. Both effects depend on  $\Delta n_{m0}$  and annealing temperature, as reported in [22].

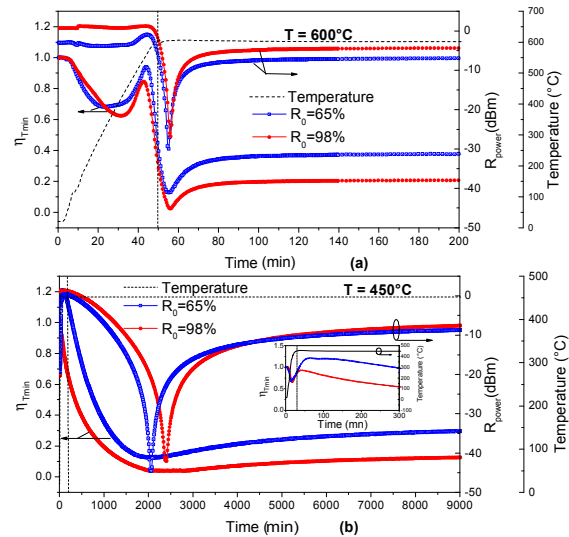


Fig. 4. Reflection power and  $\eta_{Tmin}$  for FBGs with  $R_0 = 65\%$  and  $R_0 = 98\%$  during the annealing cycles at 600 °C (a) and 450 °C (b).

The total refractive index change involves different independent effects with different relative contributions [12,25]. These contributions decay at different absolute temperatures and at different times because of their own individual activation energy distribution [21,24,26]. Moreover, the same contribution at a given temperature  $T1$  will take the required time to reach the activation energy to be involved in the process at another temperature  $T2 < T1$  [4,25]. Therefore, as discussed before, the mechanism linked to the decrease in the FBGs reflectivities is mainly due to thermal decay of the UV-induced refractive index change, which includes both color center annealing and stress effects. The last process takes more time to contribute at low temperature annealing cycle because of its high activation energy distribution (higher than that of color centers) [25]. Consequently more time (slower process) is

required for FBGs with higher concentration of stress to regenerate when temperature is lowered. Therefore in the slow regeneration regime, FBGs with higher reflectivity regenerate later than the weaker ones.

Fig. 5 (a) depicts temperature calibration and stability test of the grating at high temperature. The regenerated FBGs show the same temperature behavior with the same sensitivities:  $(12.99 \pm 0.18) \text{ pm}/^\circ\text{C}$  in the range  $120^\circ\text{C}$ - $900^\circ\text{C}$  and  $(9.66 \text{ pm}/^\circ\text{C})$  in the range  $20^\circ\text{C}$ - $120^\circ\text{C}$ . They erase when the temperature exceeds  $1000^\circ\text{C}$ . We assume that this temperature corresponds to the glass softening one, since it is lower in doped fiber ( $900$ - $1000^\circ\text{C}$ ) than in pure silica glass [24]. For seed FBGs, the wavelength shift versus temperature is not linear. This effect is due to the negative contribution of UV-induced refractive index change (thermal decay of DC refractive index of the grating starting at low temperature) and to the positive influence due to the thermal annealing of boron-related refractive index of the fiber involved at higher temperature ( $\sim 400^\circ\text{C}$ ).

Low temperature regeneration process has the advantage of preserving the mechanical strength of the optical fiber [28]. Nevertheless, at  $600^\circ\text{C}$  and even lower at  $450^\circ\text{C}$ , the acrylate coating of the fiber cannot resist. One of the possible solutions is to use reduced size oven to anneal the fiber at the location of the grating. Another potential solution is to use metal coating as proposed in [15]. As a last step in our experimental work, longitudinal strain calibration test was conducted on both seed and regenerated gratings. Figure 5b reports the experimental results. The sensitivities in the investigated strain range are almost the same  $(1.22 \pm 0.05) \text{ pm}/\mu\epsilon$  for seed and FBGs regenerated at  $450^\circ\text{C}$  and  $600^\circ\text{C}$ . Nevertheless, the breaking point for the FBGs regenerated at  $600^\circ\text{C}$  was found to be  $\sim 2500 \mu\epsilon$  while the one for FBGs regenerated at  $450^\circ\text{C}$  gratings is more than  $3500 \mu\epsilon$ .

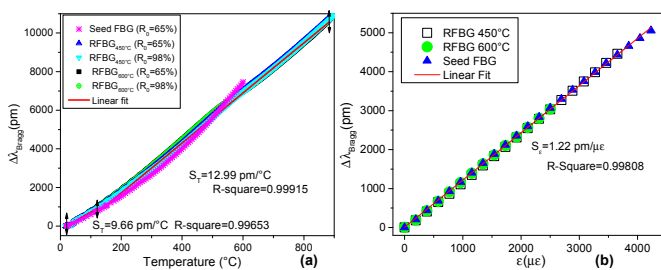


Fig. 5. (a) Temperature calibration curve of regenerated FBG at  $450^\circ\text{C}$  (b) Strain calibration curve of Seed and RFBG at  $450^\circ\text{C}$ .

In this work, we investigated the annealing temperature effect on FBGs produced in B/Ge co-doped silica fiber of different strengths. Both fast annealing cycle at high temperature ( $>600^\circ\text{C}$ ) and slow annealing one at lower temperature ( $450^\circ\text{C}$ ) have been experienced. A regeneration cycle at  $450^\circ\text{C}$  within 150 hours has been demonstrated. Moreover, at this relatively low temperature, the weakest FBG regenerates faster than the higher ones while the opposite occurs for the high temperature cycle. These behaviors as well as changes in gratings strength before regeneration cycle suggest the competition between two main mechanisms in the annealing and regeneration process: boron-annealing induced refractive index changes of the fiber and UV-induced refractive

index change of the grating including colour center annealing and stress-induced compactness of the fiber core. Temperature sensitivity of RFBG was determined to be  $(12.99 \pm 0.18) \text{ pm}/^\circ\text{C}$  in the range  $100$ - $900^\circ\text{C}$  and longitudinal strain sensitivity is equal to  $(1.17 \pm 0.01) \text{ pm}/\mu\epsilon$ .

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## References

1. B. Lee, Opt. Fiber Technol., 9(2), 57-79 (2003).
2. G. Laffont, R. Cotillard, N. Roussel, R. Desmarchelier, and S. Rougeault, Sensors, 18, 1791(2018), doi:10.3390/s18061791
3. S. J. Mihailov, Sensors (Basel) 12(2), 1898–1918 (2012)
4. T. Erdogan, V. Mizrahi, P. J. Lemaire, and D. Monroe, J.Appl. Phys. 76(1), 73–80 (1994).
5. Groothoff, and J. Canning, Opt. Lett., 29(20), 2360-2362 (2004).
6. G. Brambilla, and H. Rutt, Appl. Phys. Lett., 80(18), 3259-3261 (2002).
7. M. Fokine, J. Opt. Soc. Am. B, 19(8), 1759-1765 (2002).
8. H. Chikh-Bled, K. Chah, Á. González-Vila, B. Lasri, C. Caucheteur, Opt. Lett. 41 (17), 4048–4051 (2016).
9. S. Bandyopadhyay, J. Canning, M. Stevenson, and K. Cook, Opt. Lett., 33(16), 1917–1919 (2008).
10. E. Lindner, C. Chojetzki, S. Brückner, M. Becker, M. Rothhardt, and H. Bartelt, Opt. Express, 17(15), 12523–12531 (2009).
11. K. Cook, L. Y. Shao, and J. Canning, Opt. Mater. Express 2(12), 1733–1742 (2012).
12. J. Canning, M. Stevenson, S. Bandyopadhyay, and K. Cook, Sensors, 8 (10), 6448–6452 (2008).
13. J. Canning, S. Bandyopadhyay, M. Stevenson, P. Biswas, J. Fenton, and M. Aslund, J. Eur. Opt. Soc. 4, 09052 (2009).
14. J. Canning, S. Bandyopadhyay, P. Biswas, M. Aslund, M. Stevenson, and K. Cook, pp. 363–384, in Frontiers in Guided Wave Optics and Optoelectronics, Ed. Bishnu Pal, INTECH, ISBN 978–953–7619–82–4, February (2010)
15. D. Barrera, V. Finazzi, J. Villatoro, S. Sales, and V. Pruneri, IEEE Sens. J. 12(1), 107–112 (2012).
16. P. Holmberg, F. Laurell, and M. Fokine, Opt. Express 23(21), 27520–27535 (2015).
17. M. Celikin, D. Barba, B. Bastola, A. Ruediger, and F. Rosei, Opt. Express 24(19), 21897–21909 (2016).
18. A. Bueno, D. Kinet, P. Mégret, and C. Caucheteur, Opt. Lett. 38(20), 4178–4181 (2013).
19. H. Patrick, S. L. Gilbert, A. Lidgard, and M. D. Gallagher, J. Appl. Phys. 78(5), 2940–2945 (1995).
20. L. Polz, Q. Nguyen, H. Bartelt, and J. Roths, Optics Comm. 313, 128–133 (2014).
21. S. Pal, Opt. Commun. 262(1), 68–76 (2006).
22. K. E. Chisholm, K. Sugden, and I. Bennion, J. Phys. D Appl. Phys. 31(1), 61–64 (1998).
23. I. Camlibel, D. A. Pinnow and F. W. Dabby, Appl. Phys. Lett. 26 (4), 1183-1185 (1975).
24. W. Liu, K. Cook, and J. Canning, Sensors, 15, 20659-20677 (2015).
25. G. Violakis, and H. G. Limberger, Opt. Mater. Express 4(3), 499–508 (2014).
26. S. Pal, J. Mandal, T. Sun, and K. T. V. Grattan, Appl. Opt. 42(12), 2188–2197 (2003).
27. S. Bandyopadhyay, J. Canning, P. Biswas, M. Stevenson, and K. Dasgupta, Opt. Express, 19(2), 1198-1206 (2011).
28. T. Wang, L. Y. Shao, J. Canning, and K. Cook, opt. Lett. 38(3), 247-249 (2013).

## References

1. B. Lee, "Review of the present status of optical fiber sensors," *Opt. Fiber Technol.*, 9(2), 57-79 (2003).
2. G. Laffont, R. Cotillard, N. Roussel, R. Desmarchelier, and S. Rougeault, "Temperature Resistant Fiber Bragg Gratings for On-Line and Structural Health Monitoring of the Next-Generation of Nuclear Reactors," *Sensors*, **18**, 1791(2018), doi:10.3390/s18061791 (<https://doi.org/10.3390/s18061791>)
3. S. J. Mihailov, "Fiber Bragg grating sensors for harsh environments," *Sensors (Basel)* 12(2), 1898–1918 (2012)
4. T. Erdogan, V. Mizrahi, P. J. Lemaire, and D. Monroe, "Decay of ultraviolet-induced fiber Bragg gratings," *J. Appl. Phys.* 76(1), 73–80 (1994).
5. Groothoff, and J. Canning, "Enhanced type IIA gratings for high temperature operation," *Opt. Lett.*, 29(20), 2360-2362 (2004).
6. G. Brambilla, and H. Rutt, "Fiber Bragg gratings with enhanced thermal stability," *Appl. Phys. Lett.*, 80(18), 3259-3261 (2002).
7. M. Fokine, "Formation of thermally stable chemical composition gratings in optical fibers," *J. Opt. Soc. Am. B*, 19(8), 1759-1765 (2002).
8. H. Chikh-Bled, K. Chah, Á. González-Vila, B. Lasri, C. Caucheteur, "Behavior of femtosecond laser-induced eccentric fiber Bragg gratings at very high temperatures," *Opt. Lett.* 41 (17), 4048–4051 (2016).
9. S. Bandyopadhyay, J. Canning, M. Stevenson, and K. Cook, "Ultrahigh-temperature regenerated gratings in boron-codoped germanosilicate optical fiber using 193 nm," *Opt. Lett.*, 33(16), 1917–1919 (2008).
10. E. Lindner, C. Chojetzki, S. Brückner, M. Becker, M. Rothhardt, and H. Bartelt, "Thermal regeneration of fiber Bragg gratings in photosensitive fibers," *Opt. Express*, 17(15), 12523–12531 (2009).
11. K. Cook, L. Y. Shao, and J. Canning, "Regeneration and helium: Regenerating Bragg gratings in helium-loaded germanosilicate optical fibre," *Opt. Mater. Express* 2(12), 1733–1742 (2012).
12. J. Canning, M. Stevenson, S. Bandyopadhyay, and K. Cook, "Extreme silica optical fiber gratings," *Sensors*, 8 (10), 6448–6452 (2008).
13. J. Canning, S. Bandyopadhyay, M. Stevenson, P. Biswas, J. Fenton, and M. Aslund, "Regenerated gratings," *J. Eur. Opt. Soc.* 4, 09052 (2009).
14. J. Canning, S. Bandyopadhyay, P. Biswas, M. Aslund, M. Stevenson, and K. Cook, "Regenerated Fibre Bragg Gratings," pp. 363–384, in *Frontiers in Guided Wave Optics and Optoelectronics*, Ed. Bishnu Pal, INTECH, ISBN 978–953–7619–82–4, February (2010)
15. D. Barrera, V. Finazzi, J. Villatoro, S. Sales, and V. Pruneri, "Packaged optical sensors based on regenerated fiber bragg gratings for high temperature applications," *IEEE Sens. J.* 12(1), 107–112 (2012).
16. P. Holmberg, F. Laurell, and M. Fokine, "Influence of pre-annealing on the thermal regeneration of fiber Bragg gratings in standard optical fibers," *Opt. Express* 23(21), 27520–27535 (2015).
17. M. Celikin, D. Barba, B. Bastola, A. Ruediger, and F. Rosei, "Development of regenerated fiber Bragg grating sensors with long-term stability," *Opt. Express* 24(19), 21897–21909 (2016).
18. A. Bueno, D. Kinet, P. Mégret, and C. Caucheteur, "Fast thermal regeneration of fiber Bragg gratings," *Opt. Lett.* 38(20), 4178–4181 (2013).
19. H. Patrick, S. L. Gilbert, A. Lidgard, and M. D. Gallagher, "Annealing of Bragg gratings in hydrogen - loaded optical fiber," *J. Appl. Phys.* 78(5), 2940–2945 (1995).
20. L. Polz, Q. Nguyen, H. Bartelt, and J. Roths, "Fiber Bragg gratings in hydrogen-loaded photosensitive fiber with two regeneration regimes," *Optics Comm.* 313, 128–133 (2014).
21. S. Pal, "Characterization of thermal (in)stability and temperature-dependence of type-I and type-IIA Bragg gratings written in B-Ge co-doped fiber," *Opt. Commun.* 262(1), 68–76 (2006).
22. K. E. Chisholm, K. Sugden, and I. Bennion, "Effects of thermal annealing on Bragg fibre gratings in boron/germania co-doped fibre," *J. Phys. D Appl. Phys.* 31(1), 61–64 (1998).
23. I. Camlibel, D. A. Pinnow and F. W. Dabby, "Optical ageing characteristics of borosilicate clad fused silica core fiber optical waveguides," *Appl. Phys. Lett.* 26 (4), 1183-1185 (1975).
24. W. Liu, K. Cook, and J. Canning, "Ultrahigh-Temperature Regeneration of Long Period Gratings (LPGs) in Boron-Codoped Germanosilicate Optical Fibre," *Sensors*, 15, 20659-20677 (2015).
25. G. Violakis, and H. G. Limberger, "Annealing of UV Ar+ and ArF excimer laser fabricated Bragg gratings: SMF-28e fiber," *Opt. Mater. Express* 4(3), 499–508 (2014).
26. S. Pal, J. Mandal, T. Sun, and K. T. V. Grattan, "Analysis of thermal decay and prediction of operational lifetime for a type I boron-germanium codoped fiber Bragg grating," *Appl. Opt.* 42(12), 2188–2197 (2003).
27. S. Bandyopadhyay, J. Canning, P. Biswas, M. Stevenson, and K. Dasgupta, "A study of regenerated gratings produced in germanosilicate fibers by high temperature annealing," *Opt. Express*, 19(2), 1198-1206 (2011).
28. T. Wang, L. Y. Shao, J. Canning, and K. Cook, "Temperature and strain characterization of regenerated gratings" *opt. Lett.* 38(3), 247-249 (2013).