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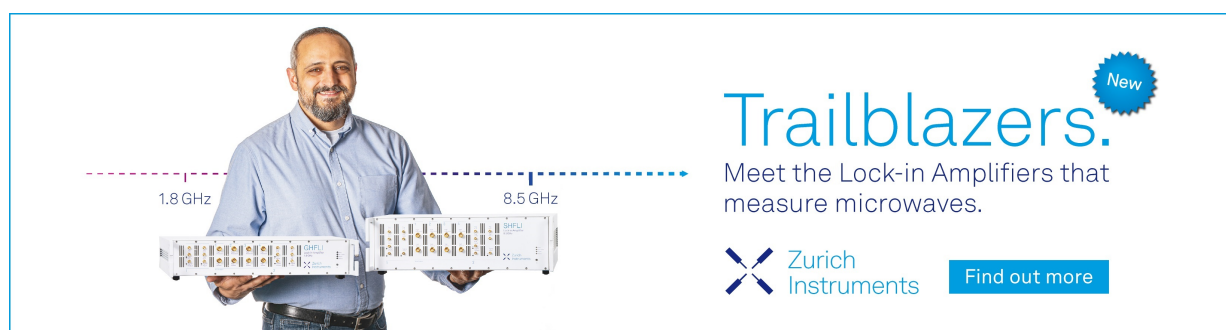
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ABSTRACT

The newly synthesized two-dimensional polyaniline (C_3N) is structurally similar to graphene and has interesting electronic, magnetic, optical, and thermal properties. Motivated by the fact that point defects in graphene give rise to interesting features, like magnetization in an all carbon material, we perform density functional theory calculations to investigate vacancy and Stone–Wales type point defects in monolayer C_3N . We compare and contrast the structural, electronic, and magnetic properties of these defects with those in graphene. While monovacancies and Stone–Wales defects of C_3N result in reconstructions similar to those in graphene, divacancies display dissimilar geometrical features. Different from graphene, all vacancies in C_3N have metallic character because of altered stoichiometry; those that have low-coordinated atoms have finite magnetic moments. We further investigate the robustness of the reconstructed structures and the changes in the magnetic moments by applying tensile and compressive biaxial strain. We find that, with the advantage of finite bandgap, point defects in C_3N are qualified as good candidates for future spintronics applications.

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I. INTRODUCTION

Two-dimensional (2D) materials are promising candidates for next-generation devices as they bring a wide range of unique properties.^{1–6} An important advantage of these materials is the ease to modify their properties by nano-structuring schemes, some of which are creating heterostructures by stacking different types of monolayers, reducing their dimensions to make ribbons or quantum dots, adsorbing functional groups to tailor their properties.^{7–10} Defects are major ingredients that determine the material properties. Defects and defect engineering in graphene related materials have been subjects of intense research during last few years.^{11,12} Monovacancy (MV), divacancy (DV), and Stone–Wales (SW) defects are the most common defect types in graphene, where MV defect is a source of magnetism and lies at the heart of graphene spintronics studies.^{13–18,29}

Two-dimensional C_3N (polyaniline) was first studied theoretically^{19,20} and was synthesized recently by Mahmood *et al.*²¹ The atoms of the C_3N monolayer are arranged in a honeycomb lattice structure like in graphene, and it is an indirect bandgap

semiconductor.²⁰ It is synthesized by direct pyrolysis of an organic single crystal and fabricated by polymerization of 2,3-diaminophenazine.^{21,22} Its electronic,^{23,27} mechanical,^{23,28} thermal,^{30,31} and magnetic^{25,26,45,46,48} properties have been studied and some possible applications were suggested such as the anode material for batteries,³² photocatalytic,³⁵ and nanosensor applications.^{24,36–38} In addition, its zigzag nanoribbons have also been studied.^{39–44}

In the present study, we perform first-principles calculations to investigate the structural, electronic, and magnetic properties of MV, DV, and SW defects on monolayer C_3N . We compare and contrast the structural, electronic, and magnetic properties of C_3N with graphene. Since it is a newly synthesized material, the properties of point defects in C_3N are not studied experimentally yet and computational investigations are expected to guide future experimental observations.

We show that, while having similar structural features, C_3N -MV structures give rise to partially filled bands and have finite magnetic moments, whereas C_3N -DV can structurally be

distinguished from their graphene counterparts and induce magnetic moments unlike in graphene. We also investigate the effects of tensile and compressive biaxial strain on defected polyaniline.

II. COMPUTATIONAL METHODS

Point defects are modeled using the cluster approach or periodic boundary conditions, both of which bring in certain spurious effects. The main shortcoming of the cluster approach is the finite size effect and interaction with the edge states. Using periodic boundary conditions, one prevents edge states but includes inter-defect couplings. Using large enough super cells, inter-defect coupling can be minimized, which is the method we choose. We note that some quantitative features such as the defect induced bandgap value or the exact value of the magnetic moment can be affected from the choice of the super cell size, which also determines the defect density. However, the qualitative features should be independent of the super cell size, provided that it is large enough. We also note that electronic band diagrams, as well as the density of states plots, should be interpreted in accordance with the choice of super cell size.

Calculations are performed within the framework of density functional theory (DFT) as implemented in the VASP code.^{49,50} The projector augmented wave (PAW) potentials are used with the Perdew–Burke–Ernzerhof (PBE) functionals of the generalized gradient approximation (GGA).^{51,52} Plane wave energy cutoff values of 400 eV and 500 eV are used for graphene and C₃N, respectively. The structures are relaxed until the force on each atom is < 0.001 eV/Å, and the self-consistency tolerance is set to 10⁻⁶ eV. Super cells consisting of 128 atoms in their pristine structures, which correspond to 8 × 8 × 1 and 4 × 4 × 1 super cells for graphene and C₃N, are used so that the interaction between defects is negligible. The vacuum spacing between layers is set to 10 Å. During the DFT calculations, k-point sampling is carefully examined to ensure that the calculation results are converged. For defective super cells, reciprocal space was sampled with 2 × 2 × 1 k-points using the Monkhorst–Pack scheme.⁵³ The k-point grid is chosen between 5 × 5 × 1 and 9 × 9 × 1 to calculate the density of states (DOS).

In order to investigate the stability of defective structures at elevated temperatures, we perform *ab initio* molecular dynamics (AIMD) simulations. The canonical ensemble with a Nosé–Hoover thermostat is used at 500 K. The time step used in AIMD simulations is 1 fs, and the total duration is 10 ps (10 000 steps).

The cohesive energy E_{coh} is defined as

$$E_{\text{coh}} = (n_{\text{C}}E_{\text{C}} + n_{\text{N}}E_{\text{N}} - E_{\text{tot}})/(n_{\text{C}} + n_{\text{N}}), \quad (1)$$

where n_{C} and n_{N} denote the number of C and N atoms, E_{C} and E_{N} are energies of single C and N atoms, E_{tot} is the total energy of the structure consideration. Defect formation energy is obtained as

$$E_{\text{form}} = E_{\text{tot}}^{\text{defective}} + \sum_{i=\text{C,N}} n'_i \mu_i - E_{\text{tot}}^{\text{pristine}}, \quad (2)$$

with $E_{\text{tot}}^{\text{pristine}}$ and $E_{\text{tot}}^{\text{defective}}$ being total energies of pristine and defective super cells, respectively. The number of removed atoms from

species i is denoted with n'_i . The chemical potentials $\mu_{\text{C}} = E_{\text{tot}}^{\text{Graphene}}/n_{\text{tot}}$ and $\mu_{\text{N}} = (E_{\text{tot}}^{\text{C}_3\text{N}} - n_{\text{C}}\mu_{\text{C}})/n_{\text{tot}}$ are those of single C and N atoms of the pristine structures. The magnetic stabilization energy is calculated using $\Delta E_{\text{mag}} = E_{\text{nonmag}} - E_{\text{mag}}$, where E_{nonmag} is the total energy of spin unpolarized calculation, and E_{mag} is the total energy of the spin polarized calculation.

III. RESULTS AND DISCUSSION

A. Structural properties

The pristine graphene and C₃N monolayers and their unit cells can be seen in Fig. 1. The lattice parameters are 2.47 Å and 4.86 Å for graphene and C₃N, respectively.²³ In the C₃N monolayer, both C–N and C–C bond lengths are approximately 1.40 Å, with C–N bond being slightly longer, which is in good agreement with previous theoretical results 1.404 Å and 1.403 Å.⁴⁷ The C–C bond length is 1.42 Å in graphene, for comparison. We consider three defect types: MV, DV, and SW. For graphene, these correspond to three structures, whereas for C₃N there are two possible structures for each defect. MV can be of C or N type, C₃N-MV(C) and C₃N-MV(N). Since DV and SW involve a pair of atoms, they can be (CC) or (CN) type. That is, we have six defective structures of C₃N.

We first investigate the structural aspects of gr-MV defect. Removing a carbon atom from the hexagonal lattice, the remaining structure is threefold symmetric before relaxation. Three carbon atoms belonging to the opposite sublattice with respect to the removed atom are two-coordinated. If these atoms are moved toward the vacancy site without breaking the threefold symmetry, they would have four bonds each, which is not favorable within sp²

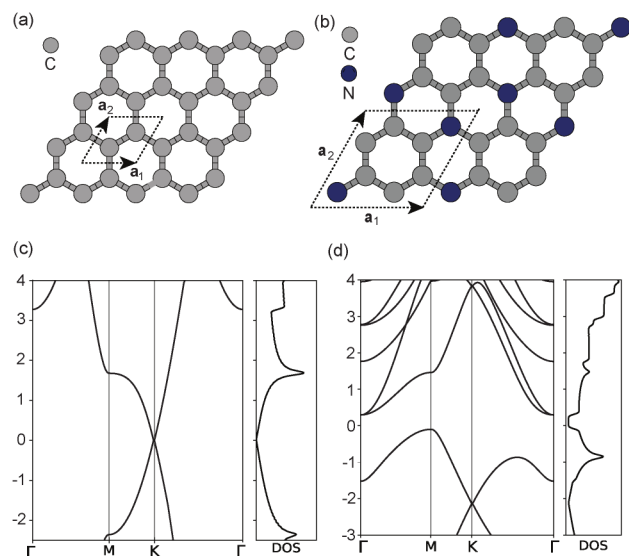


FIG. 1. Honeycomb structures of pristine graphene and C₃N monolayers are shown in (a) and (b), respectively. The unit cells are depicted with dashed lines. Electronic bands and DOS of the pristine graphene and C₃N are plotted in (c) and (d), respectively, where the zero of the energy is set to the Fermi level.

hybridization. Rather, there are three possible states which are more favorable. This is achieved by breaking the threefold symmetry and forming a pentagon–nonagon pair (5–9 reconstruction) having only one two-coordinated carbon, which is known as the Jahn–Teller distortion. [Fig. 2(a)] However, it is not trivial to obtain this structure in the simulations, which is the reason for having two different structures reported in the literature.⁵⁴ The structure needs to be perturbed to break the symmetry and the super cell should be large enough to distribute the local strain over the neighborhood. If these two conditions are not fulfilled, the simulation results in a metastable state with a threefold

symmetric geometry and three two-coordinated atoms, or even if the symmetry is slightly broken there is no new bond formation. Once the conditions are fulfilled, the structure finds the stable geometry, where two of the low-coordinated carbon atoms form a 1.82 Å long bond to stabilize. The remaining two-coordinated carbon shortens its bonds down to 1.35 Å. The formation and cohesive energies for gr-MV are 7.89 eV and 7.91 eV, respectively (see Table I) The difference in total energies between the stable and metastable structures is 23 meV. The magnetic properties of gr-MV are closely related to this reconstruction, which are discussed in Sec. III B.

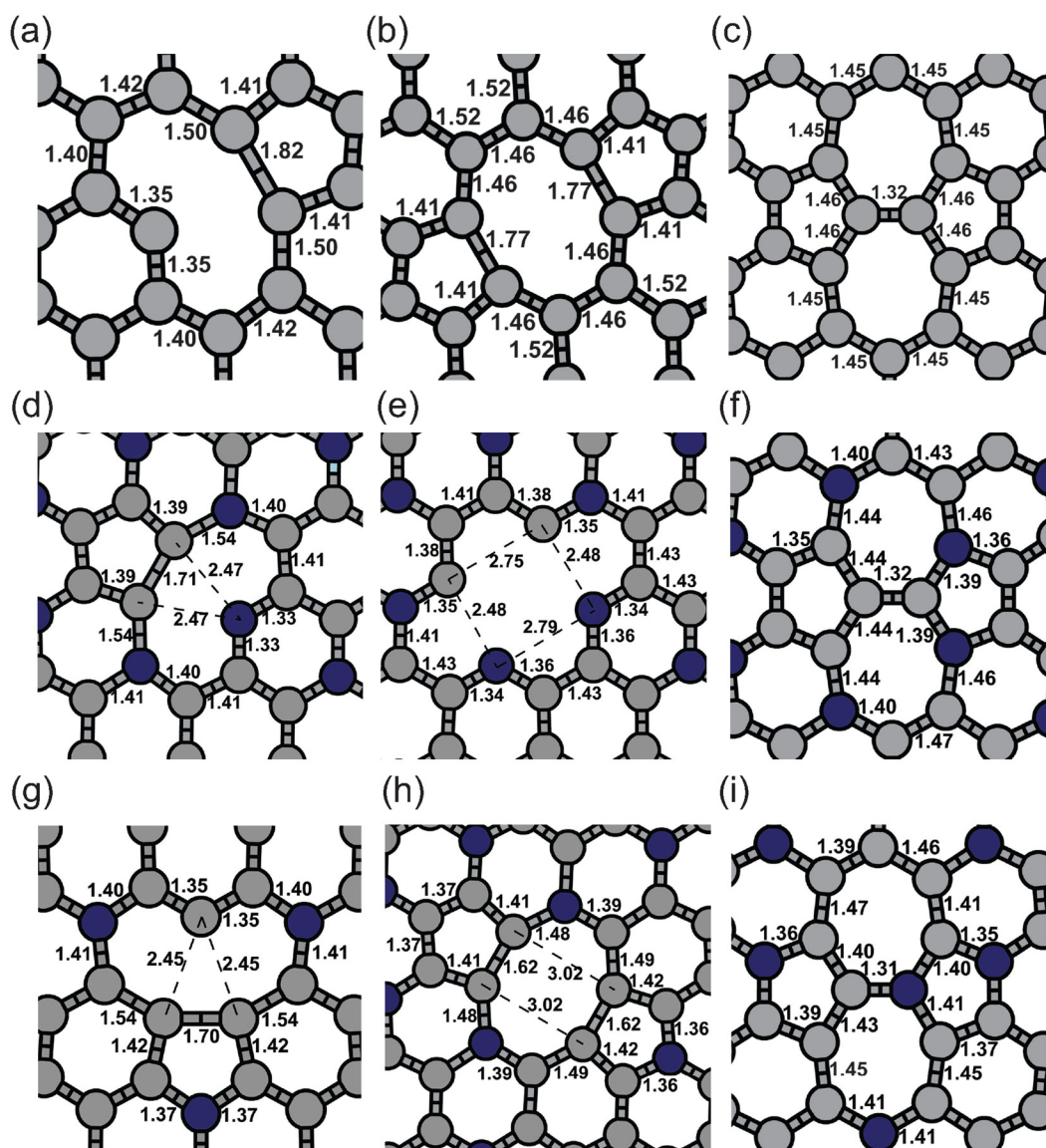


FIG. 2. Graphene and C_3N defect lattice structures (a) graphene MV (gr-MV), (b) graphene DV (gr-DV), (c) graphene Stone–Wales (gr-SW), (d) C_3N – MV(C), (e) C_3N – DV(CN), (f) C_3N – SW(CN), (g) C_3N – MV(N), (h) C_3N – DV(CN), and (i) C_3N – SW(CN).

TABLE I. Defect formation energy E_{form} , cohesive energy E_{coh} , magnetic moment m , and magnetic stabilization energy ΔE_{mag} values for different defect types. Cohesive energies of the pristine structures of graphene and C_3N are 7.98 eV and 7.08 eV for comparison.

Structure	E_{form} (eV)	E_{coh} (eV)	m (μ_B)	ΔE_{mag} (eV)
gr-MV	7.89	7.91	1.15	0.09
gr-DV	7.56	7.92
gr-SW	5.06	7.94
$\text{C}_3\text{N-MV(C)}$;	4.71	7.04	0.99	0.18
$\text{C}_3\text{N-MV(N)}$;	5.84	7.06	0.29	0.72
$\text{C}_3\text{N-DV(CC)}$;	4.90	7.03	1.23	0.16
$\text{C}_3\text{N-DV(CN)}$;	6.32	7.05
$\text{C}_3\text{N-SW(CC)}$;	5.01	7.04
$\text{C}_3\text{N-SW(CN)}$;	3.42	7.06

In $\text{C}_3\text{N-MV(C)}$ [Fig. 2(d)], there are two nitrogen and one carbon atoms that are two-coordinated; therefore, there is no three-fold symmetry as in gr-MV. Two-coordinated carbon atoms form a new bond leaving the nitrogen two-coordinated, and the 5–9 reconstruction is achieved without any need to perturb the geometry. The length of the newly formed bond is 1.71 Å in this case. The situation is similar for $\text{C}_3\text{N-MV(N)}$ [Fig. 2(g)]. This time, the initial geometry fulfills threefold symmetry, like in graphene. Perturbing the system, the symmetry is broken and a pentagon–nonagon structure is formed. The bond lengths of the reconstructed $\text{C}_3\text{N-MV(N)}$ are very close to those in $\text{C}_3\text{N-MV(C)}$. The defect formation energies for $\text{C}_3\text{N-MV(C)}$ and $\text{C}_3\text{N-MV(N)}$ are 4.71 eV and 5.84 eV, respectively. The cohesive energies are close, 7.04 eV and 7.06 eV, which agrees well with previously reported values of 6.98 eV and 7.02 eV, respectively.⁵³ Recently, Xie *et al.* have obtained different defect formation energy values. Since they have 32 atoms in 2×2 supercell of C_3N , their formation energies larger than our values.⁵⁵ The reconstructed structures of $\text{C}_3\text{N-MV}$ that we report are different from those in the literature. Ma *et al.* did not obtain any 5–9 formations for $\text{C}_3\text{N-MV(C)}$ or $\text{C}_3\text{N-MV(N)}$.⁵⁶ Rather, they have three two-coordinated atoms at the defect site, which we show to be metastable states and have differences in electronic and magnetic properties, when compared to the ground states shown here. These properties will be discussed in Sec. III B.

The divacancy structure in graphene, gr-DV, is obtained by removing a pair of neighboring carbon atoms. gr-DV has a more stable structure than gr-MV because all atoms are three-coordinated after the reconstruction.

The hexagons which lost one of their corners close to form pentagons and a pentagon–octagon–pentagon (5–8–5) geometry are obtained [Fig. 2(b)].^{57,58}

Newly formed bonds are 1.77 Å in length, similar to the bonds in gr-MV. E_{form} is 7.56 eV and lesser than that of gr-MV, while E_{coh} is 7.92 eV and higher than that of gr-MV (see Table I). In C_3N , there are two possible divacancy formations. One is removal of a carbon–carbon pair, $\text{C}_3\text{N-DV(CC)}$, the other is removal of carbon–nitrogen pair, $\text{C}_3\text{N-DV(CN)}$, as shown in Figs. 2(e) and 2(h), respectively. $\text{C}_3\text{N-DV(CN)}$ undergoes the 5–8–5 reconstruction just like in graphene, but this is not the case for $\text{C}_3\text{N-DV(CC)}$. The open hexagons do not form pentagons, and a hole with 14 edges and 4 two-

coordinated atoms (two carbon, two nitrogen) is left. We have performed lattice relaxation under biaxial compressive strain and found that even though the openings of the hexagons tend to close, there is no bonding and the 5–8–5 reconstruction does not take place even at a strain value of -0.06 . Nevertheless, E_{form} for $\text{C}_3\text{N-DV(CC)}$ is 4.90 eV, less than that of $\text{C}_3\text{N-DV(CN)}$, and E_{coh} of $\text{C}_3\text{N-DV(CN)}$ is larger than that of $\text{C}_3\text{N-DV(CC)}$, as expected.

The Stone–Wales (SW) defect is formed by rotating a bond by 90° . The number of atoms is preserved for each species. Two possible SW defects in polyaniline are $\text{C}_3\text{N-SW(CC)}$ and $\text{C}_3\text{N-SW(CN)}$. After relaxation, two pentagons and two heptagons form a 5–7–7–5 geometry for both structures considered [see Figs. 2(c), 2(f), and 2(i)]. All atoms are three-coordinated, and the rotated bond has similar lengths around 1.32 Å for all structures. Unlike the DV defect, the structural aspects of the SW defect in C_3N is not sensitive to the pair type (C–C or C–N). On the other hand, $\text{C}_3\text{N-SW(CN)}$ has considerably lower energy than $\text{C}_3\text{N-SW(CC)}$ by ~ 1.59 eV. Accordingly, its E_{coh} is larger and E_{form} is lower.

When comparing E_{form} and E_{coh} of graphene defects, gr-SW is the most stable one. gr-MV is the least stable because of the two-coordinated carbon atom. Similarly, the SW defects are among the most stable ones in C_3N , as their E_{form} are low, but there is no clear distinction between different defect types in terms of E_{form} or E_{coh} . $\text{C}_3\text{N-DV(CC)}$ has the lowest E_{coh} as it has 4 two-coordinated atoms. $\text{C}_3\text{N-MV(N)}$ and $\text{C}_3\text{N-SW(CN)}$ have the highest E_{coh} , whereas $\text{C}_3\text{N-MV(N)}$ and $\text{C}_3\text{N-DV(CN)}$ have the highest E_{form} . Namely, E_{coh} and E_{form} are not inversely related like in graphene defects, which is interpreted as a consequence of charge redistribution and unequal strain energies stored around the defect sites.

We test all the six defective structures using AIMD to see that they are stable at finite temperatures. We observe appreciable out-of-plane deformation around the defect atoms, which was absent when optimized at zero temperature. In $\text{C}_3\text{N-SW(CC)}$ and $\text{C}_3\text{N-SW(CN)}$ structures, the deformation is more pronounced than the other structures. We observe that there is no dissociation in any of the defective structures. All defects are stable at 500 K except $\text{C}_3\text{N-MV(N)}$, which is found to be metastable at this temperature. We remind that $\text{C}_3\text{N-MV(N)}$ is found to be dynamically stable according to zero temperature phonon calculations. We observe a migration of defect in the $\text{C}_3\text{N-MV(N)}$ structure for which the initial and final geometries are shown in Fig. 3. At the

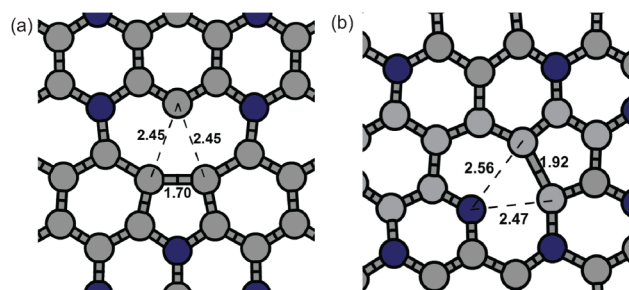


FIG. 3. Defect migration in $\text{C}_3\text{N-MV(N)}$ after the AIMD simulation. (a) $\text{C}_3\text{N-MV(N)}$ is the initial structure and (b) $\text{C}_3\text{N-MV(N)}$ is the final structure.

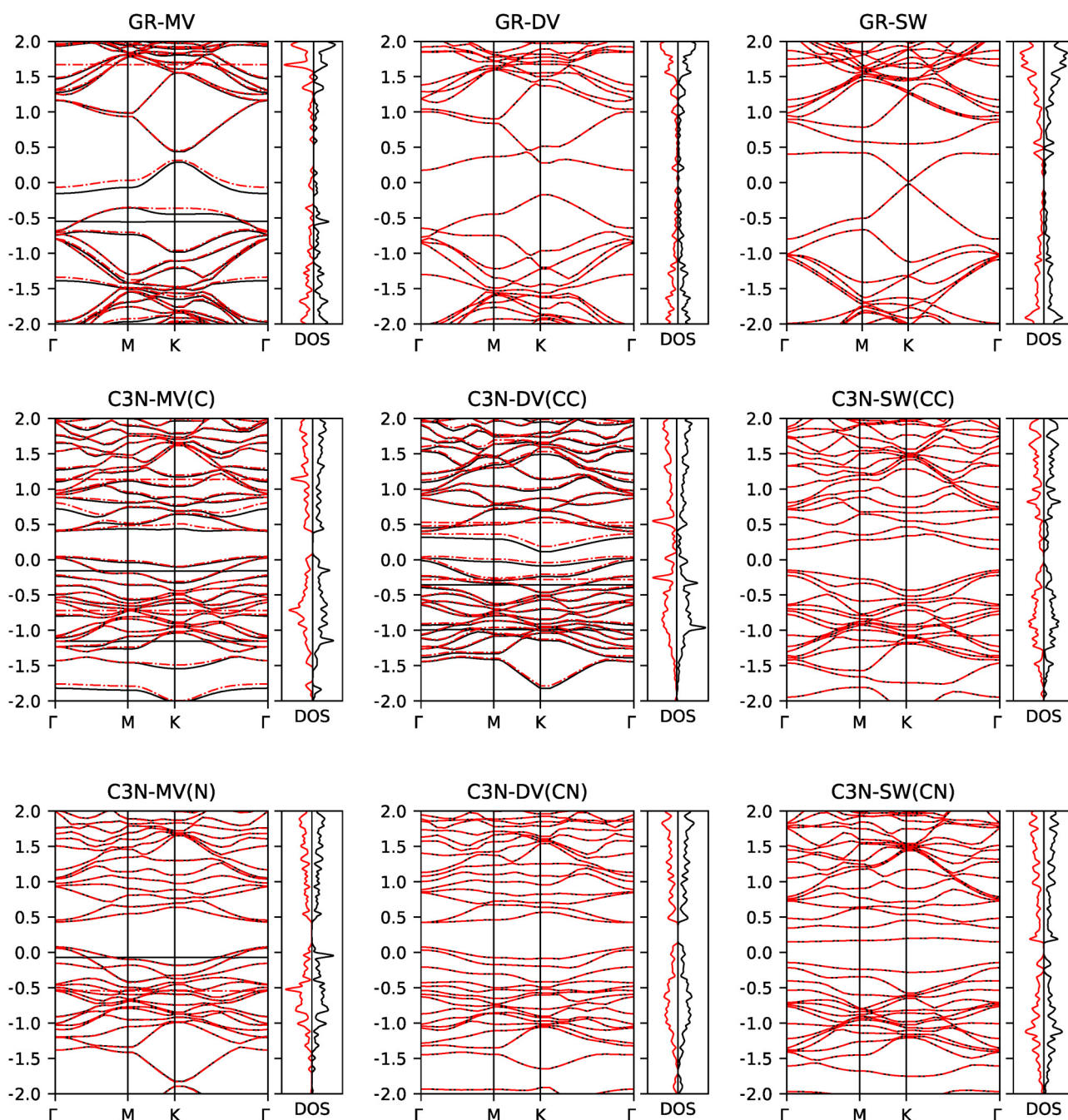


FIG. 4. Spin polarized band structure and DOS graphs of defective graphene and C_3N structures. Black-solid (red-dashed) curves represent spin-up (spin-down) electrons. The first row contains graphene structures, the remaining two are C_3N structures. Each column represents a defect type, i.e., monovacancy, divacancy, and Stone–Wales defects. The zero of the energy is set to the Fermi level.

very first steps of the AIMD simulation, two carbon atoms from the pentagon migrate to convert the nonagon to two regular hexagons, leaving the nitrogen of the pentagon two-coordinated. This final structure remains unchanged until the end of the simulation.

The total energy difference between the initial and final geometries is found to be 1.28 eV. The migration cannot be observed during the zero temperature structural relaxation process. Besides the presence of a two-coordinated nitrogen atom, the stabilized C_3N -MV

(N) structure has zero magnetic moment. This is interpreted as being due to pairing of the dangling bond with the extra electron of nitrogen.

B. Electronic and magnetic properties

Graphene is a zero bandgap semiconductor with its valence (π) and conduction (π^*) bands having linear dispersions touching at the corners of the Brillouin zone and forming Dirac cones [Fig. 1(c)].⁵⁹ On the other hand, C_3N has an indirect bandgap, with its valence band maximum at M and conduction band minimum at Γ [Fig. 1(d)]. The bandgap is predicted as 0.40 eV according to PBE calculations,^{30,38} and the corrected bandgap is 1.04 eV when HSE06 hybrid functionals are used.^{38,60,61} Corresponding band structures and DOS for defective graphene and C_3N as obtained from spin polarized DFT calculations are plotted in Fig. 4.

In graphene, π - and π^* -bands develop a finite energy gap when defects are introduced (see the first row of Fig. 4). Fermi energy lies within the gap for gr-DV and gr-SW, but it crosses the π -band in gr-MV. Moreover, there exist a flatband below the Fermi level at -0.5 eV belonging to the majority spin and another belonging to the minority spin at around 1.7 eV. Spin degeneracy is preserved and no flatbands occur in gr-DV and gr-SW structures. These findings are in agreement with the literature.^{16,17,62,63}

Nitrogen doped graphene could be viewed as an intermediate state between graphene and C_3N . Structurally, the bond lengths of the dopant atom were reported to be 1.40 Å, and almost no distortion in the planar structure of graphene was observed.^{64,65} A bandgap, whose value depends on the super cell size, is introduced at the Dirac point due to breaking of the sublattice symmetry.⁶⁶ The Fermi level is shifted to higher energies, and no magnetic moment is found, that is, the extra electron is accounted for charge doping.^{65,67} In C_3N unit cell, there are even number of electrons, and the Fermi level is inside the bandgap. Vacancies, however, alter the number of electrons in the system and electronic structures of defective systems differ, as it will be discussed below.

The gr-MV structure is arguably the most studied defect in graphene.^{16,17,54,68–76} The main reason for this is magnetization in a pure carbon system owing to the dangling bonds around the defect sites. The reason of the magnetization can be understood referring to Lieb's theorem on bipartite lattices.⁷⁷ At the dilute limit, the magnetic moment is expected to converge to $2\mu_B$, $1\mu_B$ originating from σ - and $1\mu_B$ from the π -bonds. Indeed, computations with very large super cells⁷⁸ and those employing hybrid functional methods^{16,17} have confirmed this. From our spin polarized computations having a monovacancy in an 8×8 super cell of graphene, the band structure shows a metallic character and a magnetic moment of $1.15\mu_B$ [Fig. 4(a)] with a magnetization energy of 90 meV. The non-integer magnetic moment and metallization is mainly because of the periodic boundary conditions, where the corresponding defect density is 0.0078. Our focus in this study is not the limiting cases but the main features of the defects; therefore, we do not consider very large super cells or hybrid functionals in this study.

In contrast to their similarities in geometrical reconstructions, the monovacancies in C_3N have distinctive features from gr-MV. Unlike in graphene, removing an atom (or pairs of atoms for the

cases of divacancies) from C_3N alters the stoichiometry of the super cell. This is why, not only C_3N -MV but also C_3N -DV structures exhibit metallic character, namely, partially filled bands, even when all the atoms are three-coordinated. Low-coordinated atoms give rise to a range of magnetic moments. The magnetic moment m and magnetic stabilization energy ΔE_{mag} values are given in Table I.

In C_3N -MV(C), the low-coordinated atom is nitrogen and the newly formed pentagon consists of carbon atoms, whereas in C_3N -MV(N), a carbon atom is low-coordinated and the pentagon includes a nitrogen atom. The resulting magnetic moments are $0.99\mu_B$ and $0.29\mu_B$ for C and N vacancies, respectively. The difference in magnetization is also observed in the spin density difference plots (Fig. 5), where $\Delta s = \rho_\uparrow - \rho_\downarrow$ are plotted. In C_3N -MV(C), Δs is localized on two carbon atoms, while it is located on the two-coordinated carbon atom for C_3N -MV(N). In gr-MV, all atoms around the vacancy contribute to Δs . We also note that there is higher charge density ($\rho = \rho_\uparrow + \rho_\downarrow$) around the newly formed carbon-carbon bond for C_3N -MV(N). (Fig. 6)

The C_3N -DV(CC) structure has 4 two-coordinated atoms (two carbon and two nitrogen) and a magnetic moment of $1.23\mu_B$. Δs is distributed mainly on the two-coordinated atoms, with carbon atoms being slightly more magnetized (Fig. 5). The C_3N -DV(CN) structure has zero magnetic moment, which is interpreted as being due to all atoms being three-coordinated, unlike in C_3N -DV(CC) (see Fig. 2). The C_3N -SW defects do not alter the stoichiometry and all atoms are three-coordinated; therefore, these

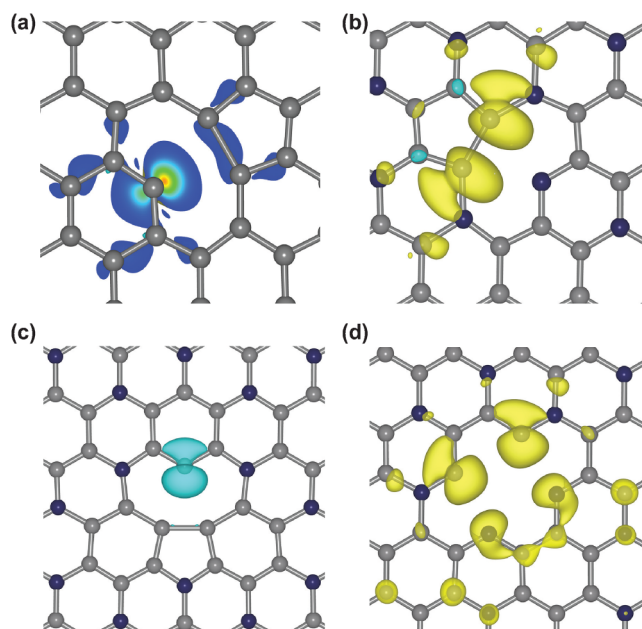


FIG. 5. Magnetization of graphene and C_3N defects. (a) gr-MV, (b) C_3N MV(C), (c) C_3N MV(N), and (d) C_3N DV(CC); the isovalue for the spin density plots ($\Delta s = \rho_\uparrow - \rho_\downarrow$) is 0.0025 e/a.u.^3 (positive values are in yellow, negative values are in blue).

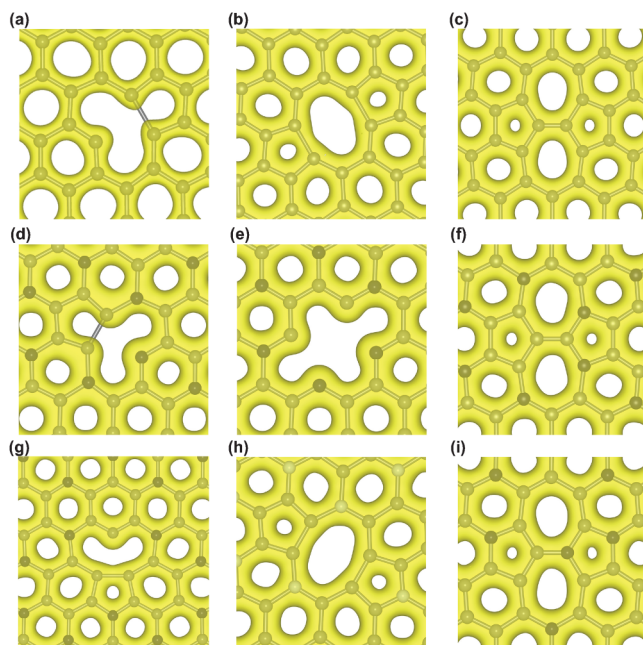


FIG. 6. Charge densities of defective graphene and C_3N structures. (a) gr-MV, (b) gr-DV, (c) gr-SW, (d) C_3N MV(C), (e) C_3N DV-C, (f) C_3N Stone-Wales-CC, (g) C_3N MV(N), (h) C_3N DV(CN), and (i) C_3N Stone-Wales-CN; the isovalue for the charge density plots is 0.075 e/a.u.^3 ($\rho = \rho_{\uparrow} + \rho_{\downarrow}$).

structures have zero magnetic moment and clear bandgaps around the Fermi energy (see Fig. 4). One should also note that for those structures with a finite magnetic moment, flatbands are observed in the electronic structure, which are absent in non-magnetic structures. For instance, in C_3N -MV(C), two spin-up flatbands appear around -0.2 eV and -1.2 eV and two spin-down flatbands appear around -0.75 eV and 1.2 eV , which are associated with the dangling bonds.

C. The effects of strain

In order to investigate the robustness of the structures and their properties, we apply tensile and compressive biaxial strain to defective C_3N structures. The magnetic moments are given in Table II for the strain between -0.04 and 0.04 . With increasing

TABLE II. Change of magnetic moments of C_3N according to applied tensile and compressive biaxial percent strains.

Strain	MV(C)	MV(N)	DV(CC)	DV(CN)	SW(CC)	SW(CN)
-4.0	0.00	0.00	0.00	0.00	0.00	0.00
-2.0	0.00	0.00	0.82	0.00	0.00	0.00
0.0	0.99	0.29	1.23	0.00	0.00	0.00
2.0	1.07	0.30	1.23	0.00	0.00	0.00
4.0	1.05	0.30	1.02	0.00	0.00	0.00

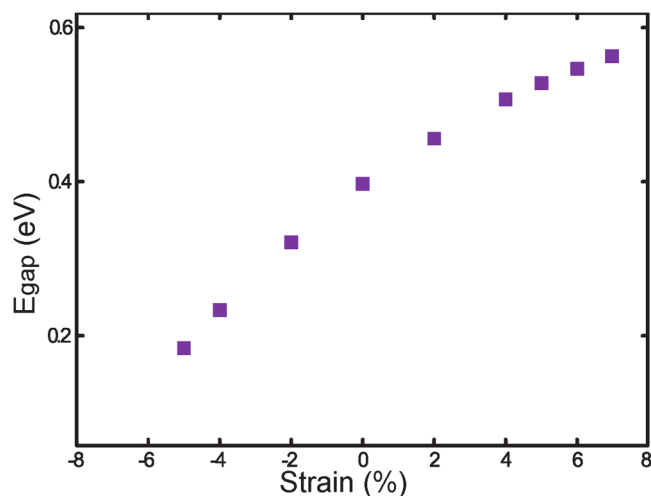


FIG. 7. Variation of the bandgap of the pristine C_3N structure as a function of applied biaxial strain.

compressive biaxial strain, total magnetic moments decrease and finally vanishes. For increasing the tensile strain, the C_3N - MV(C), C_3N - MV(N), and C_3N - DV(CC) obey the same trend. For C_3N - DV(CN) and C_3N - SW defects, applying tensile and compressive strain do not induce any magnetic moments within the tabulated range of strains. When tensile strain is increased even further, a finite magnetic moment of $0.36 \mu_B$ is observed for C_3N -DV(CN), which is associated with the broken bonds at the defect site.

We have also computed the variation of bandgap of pristine C_3N under biaxial strain. As it is shown in Fig. 7, the energy bandgap increases with increasing tensile strain, whereas the compressive strain decreases it down to 0.2 eV . We note that there is no indirect to direct bandgap transition under the tensile or compressive biaxial strain.

In order to investigate the stability of defective structures under strain, we perform AIMD for selected structures, namely, C_3N -DV(CC) and C_3N -MV(C). We note that both structures have finite magnetic moments. Applying 3.0% tensile strain, we perform AIMD simulations at 500 K. We find that both structures are stable during the entire simulation of 10 000 steps.

IV. CONCLUSION

Structural, electronic, and magnetic properties of different point defects such as monovacancy, divacancy and Stone-Wales defects on the C_3N monolayer are investigated by density functional theory. The results are compared and contrasted with similar defect structures in graphene. It is shown that some defect types [C_3N -MV(C), C_3N -MV(N), and C_3N -DV(CC)] give rise to magnetization, whereas spin degeneracy is not broken in C_3N -DV(CN), C_3N -SW(CC), and C_3N -SW(CN) defects. Magnetization is directly related to lattice reconstructions, and their values vary with the applied strain. Vacancies in C_3N could be interesting for spintronics applications and could have advantages over graphene due

to its finite bandgap. It should be possible to create defects in C_3N using AC-TEM at around 80 kV like in graphene or at smaller energies because defect formation and cohesive energies are smaller than in graphene.

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DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding author upon reasonable request.

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