

**FLEXIBLE TRANSPARENT CONDUCTING  
ELECTRODES BASED ON SILVER NANOWIRE,  
GRAPHENE, AND TWO-DIMENSIONAL  
TRANSITION METAL DICHALCOGENIDE**

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**by  
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## ABSTRACT

### FLEXIBLE TRANSPARENT CONDUCTING ELECTRODES BASED ON SILVER NANOWIRE, GRAPHENE, AND TWO-DIMENSIONAL TRANSITION METAL DICHALCOGENIDE

In recent years, transparent conductive electrodes have attracted great interests owing to their critical applications in various optoelectronic devices, such as light emitting diodes, solar cells, liquid crystal displays, optical modulators, and touch screens. In this thesis, graphene-silver nanowires-transition metal dichalcogenide based hybrid transparent and conductive electrodes have been fabricated. In order to reach this goal; (1) single layer graphene on copper foil has been synthesized in large area in a CVD furnace, (2) ultrathin and very long silver nanowires have been synthesized by using wet chemistry methods, (3) MoS<sub>2</sub> and WS<sub>2</sub> single layer flakes and multilayer thin films have been synthesized in a CVD furnace, (4) electrodes of graphene, silver nanowires, and transition metal dichalcogenides have been fabricated on rigid and flexible substrates.

## ÖZET

### GÜMÜŞ NANOTEL, GRAFEN VE İKİ-BOYUTLU GEÇİŞ METAL DİKALKOJENİD TEMELLİ ESNEK ŞEFFAF İLETKEN ELEKTRODLAR

Son yıllarda, şeffaf iletken elektrodlar, ışık yayan diyotlar, güneş hücreleri, sıvı Kristal ekranlar, optic modülatörler ve dokunmatik ekranlar gibi çeşitli optoelektronik cihazlardaki kritik uygulamaları nedeniyle büyük ilgi gördü. Bu tezde grafen-gümüş nanoteller-geçiş metal dikalkojenit esaslı hibrit şeffaf ve iletken elektrotlar üretilmiştir. Bu hedefe ulaşmak için; (1) Bakır folyo üzerine tek katmanlı grafen bir CVD fırınında geniş bir alanda sentezlenmiş, (2) ultra ince ve çok uzun gümüş nanoteller ıslak kimya yöntemleri kullanılarak sentezlenmiştir, (3) MoS<sub>2</sub> ve WS<sub>2</sub> tek katmanlı pullar ve çok katmanlı ince filmler bir CVD fırınında sentezlenmiştir, (4) grafen elektrotları, gümüş nanoteller ve geçiş metali dikalkojenidleri sert ve esnek substratlar üzerinde imal edilmiştir.

*To my dear family*

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# CHAPTER 1

## INTRODUCTION

### 1.1. Graphene

In 2004, graphene, for the first time, was shown experimentally as a new two-dimensional material and it became a rising star for many research areas due to its extraordinary mechanical, optical, and electrical properties. The first experiments were done by exfoliating the graphite with a scotch tape until achieving the one atom thickness. (K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos 2016) Graphene is a  $sp^2$  hybridized new class of material and its name comes from a graphite because it is one atom thick graphite. Graphene has monolayer packed carbon atoms in a honeycomb lattice structure and it has two carbon atoms in a single unit cell. Unlike graphene, graphitic honeycomb latticed materials have also other dimensionalities like 0D fullerenes, 1D nanotubes or 3D graphite.

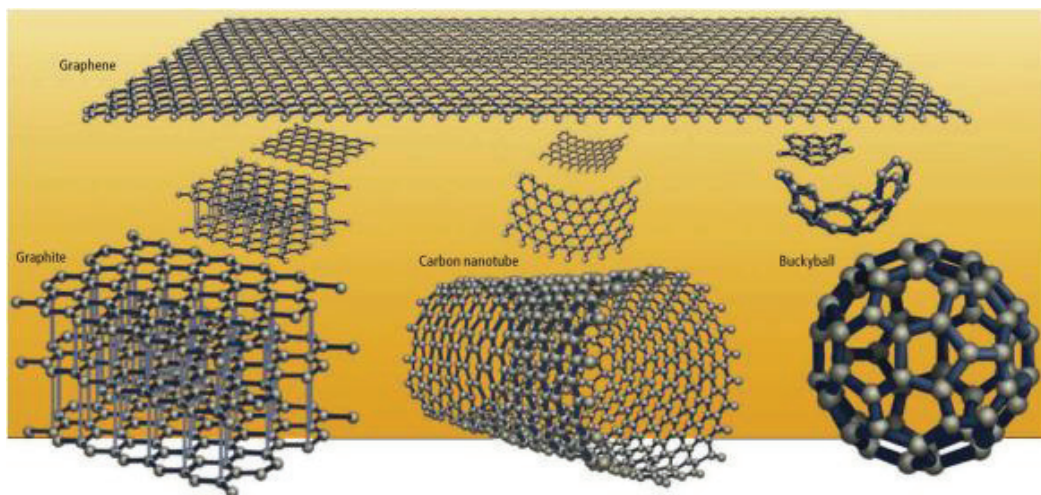


Figure 1. The basic unit of carbon nanotube, buckyball, and graphite is graphene having hexagonal latticed carbon atoms. (Source : J. Wang et al. 2017)  
The carbon atoms in all of them are  $sp^2$  hybridized.

Each carbon compounds have unique individual electronic and mechanical properties. For example, while the electrical conductivity of fullerene is  $10^{-10} S cm^{-1}$ , electrical conductivity of carbon nanotubes is shape dependent.(Castro Neto et al. 2009) In the case of graphene, the electronic structure of the graphene is layer dependent and thus the electrical conductivity of graphene rapidly evolves with the increasing number of graphene layer. Monolayer and bilayer graphene have similar electronic spectra and they are zero-gap semiconductor or semi-metal. The electronic spectra of multilayer graphene change for three or more layers like appearing the several charge carriers, overlapping the conduction and valance bands.(Castro Neto et al. 2009) As a result of unique energy band structure of graphene, graphene has extraordinary electronic and optical properties. These properties lend the usage of graphene in many areas, such as field-effect transistors, memory devices, sensing platform, photovoltaic devices, flexible electronics, and also as an infrared camouflage material(Salihoglu et al. 2018) and a transparent electrode as well.(Huang et al. 2011) Most of the reported applications of graphene in various fields require changing the charge density and Fermi energy level of graphene by application of a gate voltage.

The number of electrons and holes per unit volume implies the charge density or charge carrier concentration. The optical and electronic properties of graphene are strongly influenced by the charge carrier density. The charge carrier density of graphene can be varied by using variety of techniques, for example, electrostatically(Lee et al. 2008), chemically(H. I. Wang et al. 2017) or by photo-induction(Kim et al. 2012). Changing the carrier concentration of graphene shifts the Fermi level of graphene and this shift determines doping level of graphene. There are two types of doping, negative (n) and positive (p). In the case of negative doping, the number of electrons is greater than the number of holes and the vice versa for positive doping.(Avouris 2010)

Until now, various synthesis techniques have been shown in the literature to synthesize single layer graphene. These techniques are exfoliation, thermal decomposition, un-zipping CNTs, plasma-enhanced chemical vapor deposition, and thermal chemical vapor deposition.(Choi et al. 2010)(Zhu et al. 2010) The physical properties of Graphene can be affectively tuned for a desired application by changing the mentioned methods and synthesis conditions.(Wu et al. 2018) Up until now, several characterization techniques have been used for spectroscopic characterization of graphene. However, Raman spectroscopy has been widely used as a characterization

technique for graphene owing to the easy determination of the dimensionality, the number of layers, crystal edge structure, and doping of graphene.

### 1.1.1. Raman Spectroscopy of Graphene

Raman spectroscopy discovered by C. V. Raman in 1928 is a standard, non-destructive spectroscopic technique used for crystals, semiconductors, and for molecules to analyze vibrational modes of the structures. Different from the Rayleigh Scattering where photons are elastically scattered, Raman spectroscopy, known as Raman scattering or Raman effect, depends on inelastic photon scattering. The molecular vibrations of the system is affected by the incident laser light, phonons or other excitations, causing the laser photons' energy to be moved up or down in the spectrum. The energy change offers information on the vibrational modes in the system. The Raman spectroscopy is a quite efficient technique for the characterization of graphene. The technique can reveal dimensionality, the number of layers(Malard et al. 2009), crystal edge structure(Solomon 1980), and doping.(Ferrari 2007)

In graphene, the excitation of laser causes the two main peaks in the raman spectrum; first in plane vibrational mode is G peak and the stacking order is 2D peak.(Kahl 2015) The positions of the peaks are typical for the characterization of graphene, the G peak is placed at  $1580\text{ cm}^{-1}$  and the 2D peak is placed at  $2690\text{ cm}^{-1}$ .(Saito et al. 2011) The given positions have been most of the time measured with the 532 nm excitation laser. The Raman spectrum will change with the number of graphene layers increase.(Ferrari 2007) The number of layer increment effect is also seen at the G band such as small red shift.(Malard et al. 2009) Although the 2D band of the monolayer graphene has a single sharp peak, AB-stacked the most stable form of bilayer graphene, that is overlapped half of the carbon atoms of two layers and it has a wider peak, which is composed of four peaks.(Partoens and Peeters 2006) Finally, 2D band of the multilayer graphene becomes even broader and looks like shouldered two peaks. Apart from determining the number of layers, Raman spectroscopy can also determine the doping level and defect of the graphene.(Beams, Gustavo Cançado, and Novotny 2015)

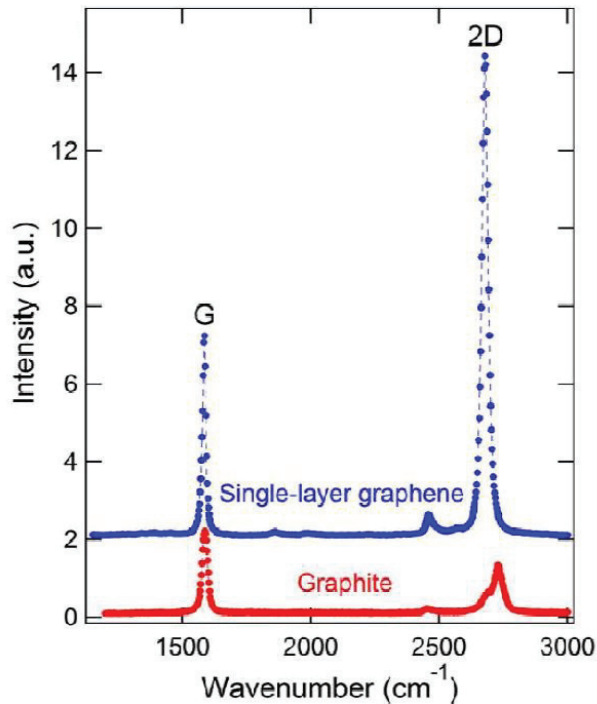


Figure 2. Raman spectra of graphene and graphite. The G and 2D bands reflecting the structural properties of the samples are clearly visible in the spectra.(Source : Kahl 2015) The relative intensities of G and 2D peaks give information about the number of graphene layers.

P-type or n-type doping changes the fermi surface of the graphene and hence the electrical and optical properties of graphene can be also changed. The existence of doping change can be identified with the Raman spectroscopy, the G peak of the graphene responds to the doping. A blueshift and narrow FWHM of the G band means that the doping exists in the graphene, but it can not give any information about doping types.(Tang, Guoxin, and Gao 2010) It also needs to be analyzed by 2D band for identifying the doping type. Characterization of electron or hole type doping can be identified with the frequency shift of the 2D peak; the upshift of the peak means the hole doping whereas the red shifts of the peak means the electron doping.(Tang, Guoxin, and Gao 2010) As a result, the G and 2D peak position is a reliable method for the identifying the type (n-type or p-type) and quantity of doping (amount of doping).(Berciaud et al. 2008)

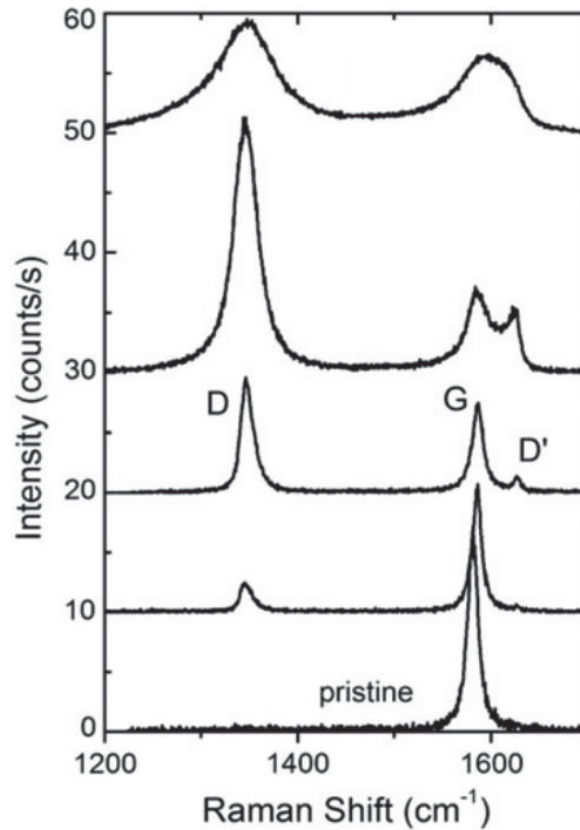


Figure 3. Raman spectra of the defective Graphene.  
 (Source : Beams, Gustavo Cançado, and Novotny 2015)

Raman spectroscopy can also identify the defect and the type of defect with the change of band intensity and FWHM.(Das, Chakraborty, and Sood 2008) The point defect is generally known as the symmetry breaking that are due to the presence of vacancies, adatoms, and substitutional atoms. Non-defective graphene has no D band, so the D band will appear when the graphene lattice have defects.(Ni et al. 2008) The defect dose is directly related to the D band intensity, the increase of the defect dose causes the D band intensity increment and thus D band becomes evident near the G band.(Dresselhaus et al. 2010) In addition, the D band is also affected by the defect dose increment, the increment seems like convolution of the G and D band. For the maximum doping of the lattice, band peaks become broadened.(Dresselhaus et al. 2010)

### 1.1.2. Electronic Properties of Graphene

An ideal hexagonal honeycomb lattice structure of graphene has two C-C atoms in a unit cell with a length of 1.42 Å. In addition to that, carbon atoms have a constant rotation angle of 120°.(Weng et al. 2018) Each lattice of graphene has three strong localized  $\sigma$  bonds, and delocalized  $\pi$  bond in the perpendicular direction. Thanks to  $\pi$  bonds of graphene, freely moving  $\pi$  electrons provide a good electrical conductivity of graphene. The honeycomb lattice includes two sublattices, as mentioned by A site and B site.(Partoens and Peeters 2006)

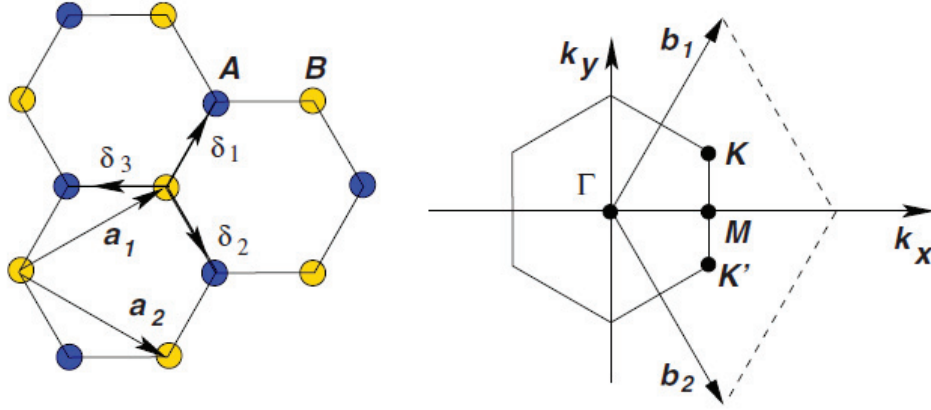


Figure 4. Honeycomb lattice of graphene and its Brillouin zone. The  $sp^2$  hybridized carbon atoms make the graphene lattice. Each graphene unit cell contains two carbon atoms.

The lattice vectors  $a_1, a_2$  and the reciprocal lattice vectors  $b_1, b_2$  can be written as

$$a_1 = \frac{a}{2} (3, \sqrt{3}) \quad \text{and} \quad a_2 = \frac{a}{2} (3, -\sqrt{3}) \quad (1.1)$$

where  $a$  is the distance between the two carbon atoms in the lattice.

$$b_1 = \frac{2\pi}{3a} (1, \sqrt{3}) \quad \text{and} \quad b_2 = \frac{2\pi}{3a} (1, -\sqrt{3}) \quad (1.2)$$

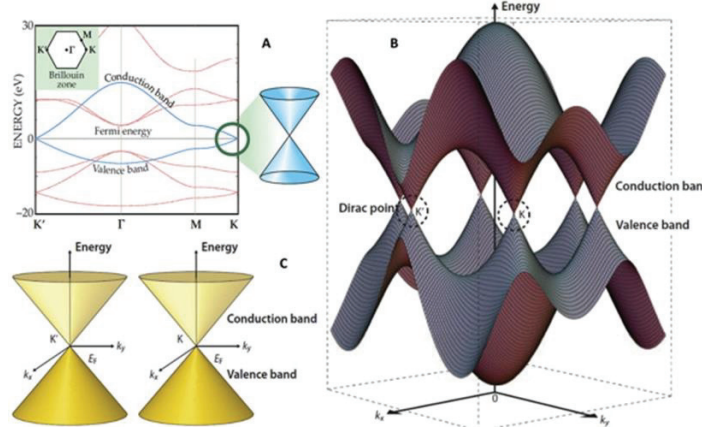


Figure 5. 2D and 3D band structure of graphene.

(Source : J. Wang et al. 2017)

The electronic band structure shows the zero energy gap of the graphene whose valance and conduction bands touch each other at two symmetric points that are  $\mathbf{K}$  and  $\mathbf{K}'$ , which is known as the Brillouin symmetry point. The positions in momentum space are

$$K = \left( \frac{2\pi}{3a}, \frac{2\pi}{3\sqrt{3}a} \right), \quad K' = \left( \frac{2\pi}{3a}, -\frac{2\pi}{3\sqrt{3}a} \right) \quad (1.3)$$

The energy of the intersection point is linearly related to the wave vector. Therefore, the massless electrons and holes in graphene, they are called Dirac fermions and the crossing point is the Dirac point.(Avouris 2010) The individual carrier properties and the Dirac Fermi characteristic of graphene lead to the Quantum Hall effect where the Hall conductance of 2D electrons show quantized conductivity.

Additionally, approximated group velocity of charge carriers is  $1 \times 10^6 \text{ m/s}$  at the Dirac point.(Morichi et al. 2008) The Fermi level of defectless, freestanding, and undoped ideal graphene is expected to be at the Dirac point.(Avouris 2010)

The square root of the carrier density is directly proportional to the Fermi energy,  $E_F \sim \sqrt{n}$ .(Castro Neto et al. 2009) Therefore, by modulating the carrier density in graphene, the active tuning of the Fermi level is possible. As mentioned above, the carrier density can be changed by doping in several ways. Moreover, electrical, and optical properties of graphene can be affected by Fermi level, so many applications of graphene are directly related to the Fermi energy modulation.



### 1.1.3. Optical Properties of Graphene

The unique optical properties of graphene come from exclusive electronic band characteristic, which is the gapless band structure. Tunability of charge carriers with high mobility, and atomic thickness of graphene provide gate tuning of the optical properties.(Yan et al. 2007) Due to one atom thickness, graphene can yield immense optical absorption thanks to interband and intraband electronic transitions in graphene band structure.(Gan, Zhang, and Bao 2017)

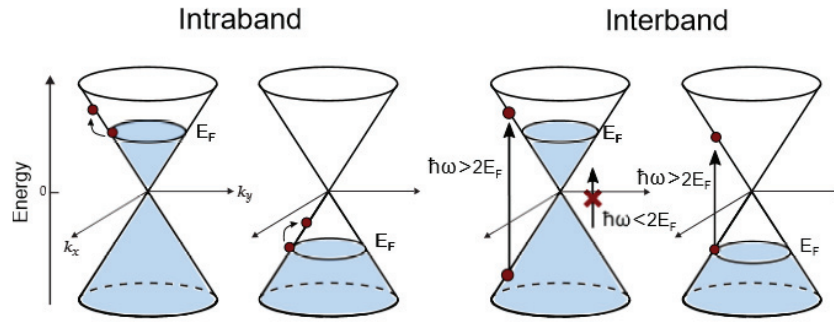


Figure 6. Intraband and interband electronic transitions in graphene.

(Source : Yakar 2020)

The intraband transition is an electronic transition in the same band. Conversely, the interband transition is a transition between the valance band and conduction band, it brings a  $\sim 2.3\%$  absorption.(Li et al. 2008) A broadband absorption of graphene occurs in a broad range of range including the visible region too. The transmittance of graphene can be written as

$$T = \left(1 + \frac{1}{2}\pi\alpha\right)^{-2} \quad (1.4)$$

where  $\alpha$  is the fine structure constant and  $\alpha = e^2/\hbar c \sim 1/137$ . (Nair et al. 2008)

The Fermi energy is important for electronic transitions, also for optical properties. For  $2E_F > E_P$ , where  $E_P$  is the incoming photon energy and  $E_F$  is Fermi energy,

interband transitions are not allowed, because of Pauli blocking.(Bao et al. 2014) If the photon energy is larger than Fermi energy such as  $E_p > 2E_F$ , interband transitions are allowed. The Fermi energy can be tuned by doping of graphene for making the interband transitions possible.(Sun et al. 2010) Consequently, the optical absorption can be altered by changing the Fermi energy.

Unlike the interband transitions, graphene has intraband transitions for  $2E_f > E_p$  situation, which is at longer wavelengths and fall into the terahertz part of the electromagnetic spectrum. The terahertz spectrum causes graphene to behave like a conductive film, and the equation of optical conductivity for graphene is given by

$$\sigma(\omega) = \frac{\sigma_{DC}(E_F)}{(1 + \omega^2\tau^2)} \quad (1.5)$$

where DC electrical conductivity is  $\sigma_{DC}(E_F)$ , and the carrier momentum scattering time is represented by  $\tau$ , and the photon frequency is the  $\omega$ . In conclusion, the optical absorption of graphene can be tuned by manipulating the conductivity or the Fermi level of graphene for the area of application.(Sun et al. 2010)

## 1.2. Silver Nanowires

Transparent conductive electrodes have recently attracted great amount of attention with their widely use in new generation electronic devices. The high cost of transparent conductive electrodes, such as commonly used indium tin oxide (ITO), has led to the need for low-cost methods. For example, ITO is a fragile material, and it restricts the application area for flexible devices.(Tseng et al. 2020) Recent experiments have shown that silver nanowires (AgNWs) are promising nanomaterial for overcoming these problems. The most important difference is diffusive transmittance and specular transmittance of AgNWs when compared with ITO. In recent years, AgNWs have received increasing attention with their use in touch screen panels(Jo et al. 2016), solar cells(Jin et al. 2018), conductive electrodes(Meng et al. 2018), flexible electronic devices(Tseng et al. 2020), and conductive thin films(D. Li et al. 2018). Good electrical conductivity and mechanical flexibility of AgNWs are striking properties of AgNWs. The Ag NWs can be synthesized with various methods such as hydrothermal

method(Chen et al. 2010), electrochemical techniques(B. S. Kim et al. 2017), wet chemical synthesis(Zhang et al. 2004), DNA template synthesis(Zhao et al. 2009), and polyol process(Trung et al. 2017). In this thesis, a wet chemical method has been used for the synthesis of uniform thin and very long AgNWs. The synthetic method was used for long and thin nanowires in order to improve the performance parameters such as sheet resistance and optical transmittance of the transparent electrodes fabricated on glass and flexible substrates. For instance, the transparent and large-area AgNW films have been developed with sheet resistances of less than 10  $\Omega$ /sq and 80%-90% transparency.(Cao et al. 2020) The AgNW density of the electrode directly affects the conductivity and transmittance of the electrode. Also, the length and diameter of AgNWs are very important parameters to improve transmittance (low) and conductivity (high).

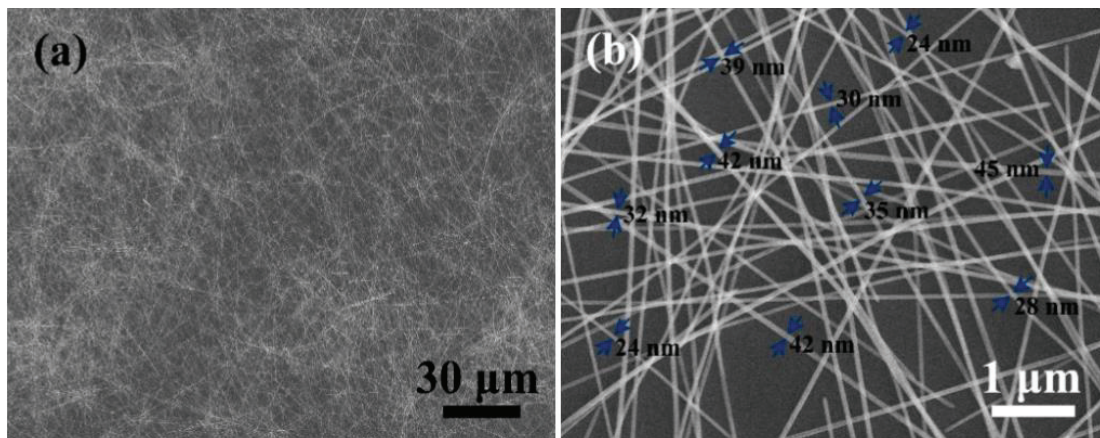


Figure 7. a) Low-magnification FE-SEM image of the synthesized AgNWs; and b) High-magnification FE-SEM image of the synthesized AgNWs.(Source : Y. Li et al. 2019)

The efficiency of the AgNW electrodes depends primarily on a bunch of features, including the structure of the nanowire such as length, diameter, size dispersity, and composition. The performance of AgNWs can be greatly improved by plasmonic welding, thermal annealing, and roll compression. These processes reduce contact resistance and increase the mechanical stability of the transparent conductive electrodes.(Cao et al. 2020)

### **1.2.1. Properties and Performance of the Silver Nanowire**

Electrical characteristic of the AgNWs is extremely important for flexible conductive electrode. Due to the high electrical and thermal conductivities of bulk silver, AgNW is a very good candidate for flexible electrode applications.(Jia, Yang, and Zhang 2014) The sheet resistance of the AgNW is an important parameter for determining the electrical characteristic. The sheet resistance depends on the junction between AgNWs, and the distribution state of AgNWs. Indeed, high aspect ratio AgNWs is desirable for high performance transparent conductive electrodes.(Mutiso et al. 2013) Long wires with less contact means the low resistivity by reducing the contact resistance.(Mutiso et al. 2013)

Device fabrication method is a significant process that affects the sheet resistance and optical transmittance of the transparent conductive electrodes. The arrangements of the AgNWs directly affect the electrode conductivity performance, because of the connection between AgNWs. Due to polymer assisted synthesis, the surface of AgNWs has polymer residues that cause higher resistivity of the resulting fabricated conductive electrodes. Thermal annealing is a useful technique for solving the polymer residue problem of the AgNWs. It has been experimentally shown that the sheet resistance of the conductive electrode has been significantly decreased after annealing of the AgNWs.(Lagrange et al. 2015)

The transparency of the conductive electrode is a very strong indicator for the applications of flexible transparent electrodes in optoelectronic devices. The critical parameter is haze factor that is the ratio of diffuse transmission to direct transmission of the light, haze factor = scattered light/transmitted light.(Langley et al. 2013) The distribution of the AgNWs on the substrate will affect light transmittance of the electrode. Also, light transmittance is directly related to density of the AgNWs. The increase of AgNW density causes a decrease in the transmittance of the incoming light.(Hu et al. 2010) In conclusion, the highest transmittance and lowest resistance are strongly needed for the ideal transparent conductive electrodes.

### 1.3. Transition Metal Dichalcogenides

Two-dimensional materials have attracted enormous attention after the discovery of graphene, which is a one-atom thick sheet of carbon. Atomically thin two-dimensional transition metal dichalcogenide materials share some similarities with graphene. The similarities are the composition of the lattice structure as a single atomic layer with hexagonal arrangement and the Van der Waals interaction between layers. The most important difference between a TMDC and graphene is the electronic band gap.(Castro Neto et al. 2009) Unlike graphene, TMDCs have a band gap and the band gap condition is changed by the layer number.(Chaves et al. 2020) While the bulk TMDCs have indirect band gap, monolayer TMDCs have direct band gap. The energy bandgaps of TMDC types range from infrared to visible energy spectrum. A bulk TMDC crystal has a bandgap of around 1.1 eV but monolayer TMDCs can further increase to 1.5-2.2 eV.(Chhowalla et al. 2013) Despite the carrier mobility of TMDC is not as good as that of graphene, the TMDCs can reach high carrier mobility levels. All of these unique electronic properties make TMDCs promising candidate for next generation atomically thin device applications.

#### 1.3.1. Crystal Structure of TMDCs

TMDCs are semiconductor materials and they can be represented generally with the chemical formula  $\mathbf{MX}_2$ , where M represents transition metal atom and X is a chalcogen atom (S, Se or Te). The layered structure of TMDC, X-M-X, is a sandwich-like structure that metal atoms are in between the two-layer of chalcogen atoms.(Chhowalla et al. 2013) The monolayer TMDC is only several Ångström thick. This reveals quantum confinement in the out-of-plane direction.(C. Lee et al. 2010) The tunable novel optical and electronic properties come from the quantum confinement, which can be tuned by an electric field, doping, and strain.(Chang et al. 2013) The surface of TMDCs is free of the dangling bonds and their structure is mechanically very robust. Moreover, Van der Waals interaction of TMDCs enables 2D materials to construct heterostructures without lattice mismatch. (Novoselov et al. 2016; Y. Liu et al. 2016; Jariwala, Marks, and Hersam 2017)

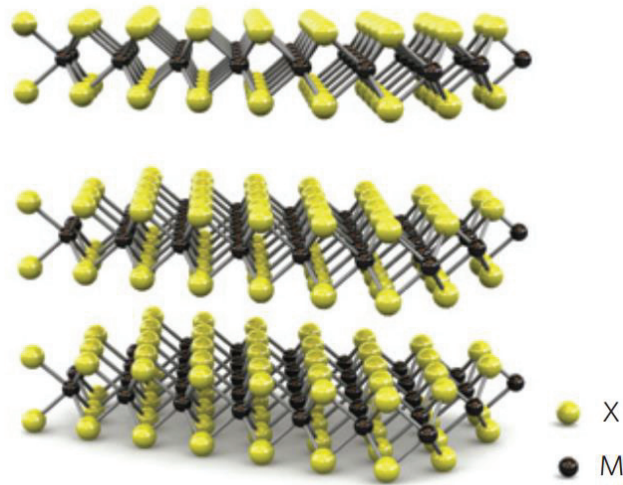


Figure 8. Crystal structure of TMDC.

(Source : Q. H. Wang et al. 2012)

The layered crystal structure arrangement of TMDC has three different types; the first is octahedral structure (1T phase), the second is trigonal prismatic structures (2H and 3R phases).(Toh et al. 2017) The crystal structure of TMDCs shows different properties: for instance, the 1T phase of TMDC reveals metallic-like properties whereas the other phases, 2H and 3R types, are semiconductors.(Datar, Bar-Sadan, and Ramasubramaniam 2020; Chia et al. 2015) Also, the application area can make use of the crystal structure of TMDC compounds. The 2H and 3R phases are more useful for electronic and optoelectronic applications with their stable crystal structure and semiconductor feature.

### 1.3.2 Electronic Band Structure of TMDCs

The electronic band structure of TMDC is highly important for tuning the optical and electronic properties. According to the evolution of thickness-dependent electrical band structure, TMDC can be used in many electronic and optoelectronic applications. The theoretical calculation of bulk 2H phase of MoS<sub>2</sub> shows the conduction band minimum is located at the middle of  $\Gamma$  and K point, and the valance band maximum

placed at the  $\Gamma$  point. Therefore, the bulk 2H phase of MoS<sub>2</sub> has an indirect band gap characteristic with a band gap of 1.2 eV.(Kuc and Heine 2015)

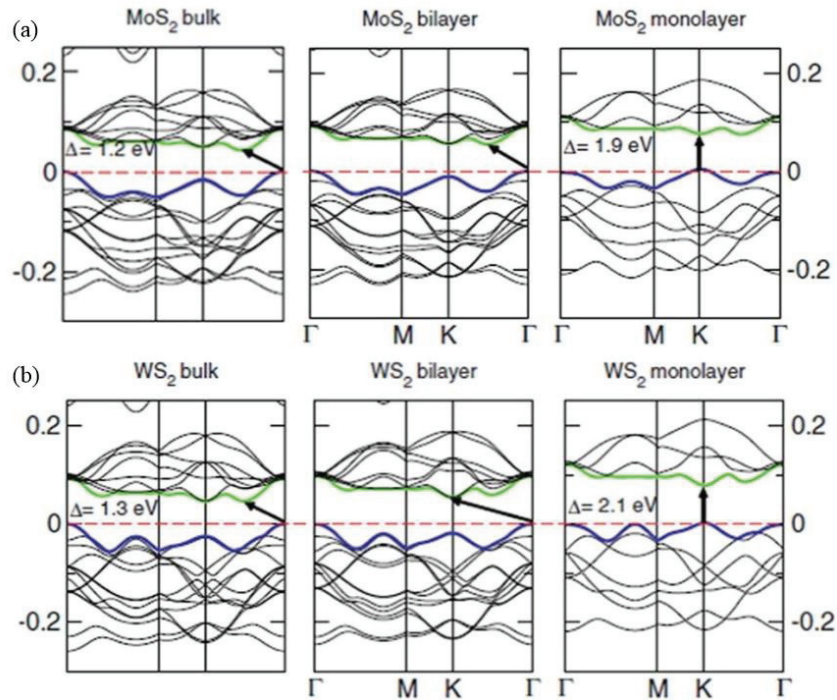


Figure 9. Electronic band structures of MoS<sub>2</sub> and WS<sub>2</sub> ; a)MoS<sub>2</sub>, b)WS<sub>2</sub>

(Source : Kuc and Heine 2015)

Theoretical calculations of the TMDC band structure show that the band structure changes as the material gets thinner. For example, when the MoS<sub>2</sub> gets thinner from bulk to monolayer, the band structure of MoS<sub>2</sub> becomes wider such as 1.2 eV to 1.9 eV.(Splendiani et al. 2010) The other TMDC materials such as WS<sub>2</sub>, MoSe<sub>2</sub>, WSe<sub>2</sub> have a similar indirect-to-direct band gap transition, when the TMDC material is monolayer. The motivations of the transition are orbital hybridization between  $P_z$  orbitals of chalcogenide atoms and the  $d$  orbitals of the transition metals and the changes in coulombic interlayer interaction.(Chang et al. 2013) The direct bandgap exhibits strong photoluminescence (PL) emission in only monolayer TMDC.(Mak et al. 2010) The difference between the binding energy of excitons and bandgap corresponds to the optical bandgap. In addition, the large binding energy maintains the TMDC in a stable state even at room temperature, including strong PL intensity.(Berkelbach, Hybertsen, and Reichman 2013) However, PL profile of the monolayer TMDC is easily affected by

the environment such as underlying substrates, and gas molecules adsorbed by the surface of the TMDC crystal. The environmental effect can be identified with doping and strain, which influence emission peak wavelength and intensity due to the manipulation of the monolayer TMDC bandgap.(Conley et al. 2013; Ross et al. 2013; Mahmoodi et al. 2016)

### **1.3.3. Characterization Methods of TMDCs**

#### **Raman Spectroscopy**

Raman spectroscopy, based on the Raman Effect, is a powerful and non-destructive technique for the characterization and analysis of TMDCs. Vibrational activity is measured for the individualization of the TMDCs by Raman active modes. Every TMDC has characteristic Raman active modes such as  $E_{2g}^1$  and  $A_{1g}$ .(Plechinger et al. 2012) The place of the mode peaks are used for identifying the TMDC, and peak amplitude is used for the estimation of the layer number. For example, MoS<sub>2</sub> has characteristic modes  $E_{2g}^1$  and  $A_{1g}$  at  $387\text{ cm}^{-1}$  and  $408\text{ cm}^{-1}$ , respectively.(Ganorkar et al., n.d.) Moreover, the distance between the two peaks denote the layer number of TMDC and that means the energy separation is used for identification of the TMDC monolayer.

#### **Photoluminescence Spectroscopy**

Photoluminescence is light emission of a material, which is a result of excited electron transition to a ground state. Photoluminescence spectroscopy is a measurement of emitted photons when these transitions occur. The radiative transitions are characteristic for each TMDC such as MoS<sub>2</sub> has a PL peak at 675.75 nm.(Ganorkar et al., n.d.) The advantage of a PL measurement is that it is non-invasive technique for determination of the optical bandgap of the material. Thanks to having direct bandgap of monolayer TMDCs, PL spectroscopy is a powerful and non-invasive tool for identifying monolayers.



## **Atomic Force Microscopy**

Atomic force microscopy is a powerful characterization technique used to determine the surface morphology and thickness. The thickness of a TMDC crystal is directly related to the layer number and the surface morphology can give information about defects of the TMDC crystal. Atomic force microscopy has two commonly used modes, tapping mode and contact mode. The difference between these modes is the position of the cantilever. In the tapping mode the distance between the cantilever and sample is more than that in the contact mode. The measurement using the contact mode is more accurate, but it can be harmful for the sample. However, tapping mode is easier, non-destructive, and sufficiently accurate.

### **1.3.4. Molybdenum Disulfide**

Molybdenum disulfide ( $\text{MoS}_2$ ) is one of the most promising TMDC nanomaterial.  $\text{MoS}_2$  has attracted great interest owing to its interesting electronic and optical properties. The physical properties  $\text{MoS}_2$  are attractive because of the layer-dependent bandgap, which can be tuned with the number of layers.(Chhowalla et al. 2013; Chaves et al. 2020)  $\text{MoS}_2$  monolayer is well-known for its electronic properties identified with the quantum confinement and surface effect. Besides, the semiconductive feature, the monolayer form has a direct bandgap that has radiative activity in the visible spectrum.(Conley et al. 2013) For instance, the energy bandgap changes from indirect ( $\sim 1.29$  eV) to direct bandgap ( $\sim 1.80$  eV) as the layer number decreases to monolayer. The direct bandgap of monolayer (1.8 eV) makes  $\text{MoS}_2$  very attractive for switching nanodevice applications.(Kuc and Heine 2015) Mainly, Atomic force microscopy, Raman spectroscopy, and photoluminescence spectroscopy have been used for spectroscopic characterization of  $\text{MoS}_2$ . The distance between Mo and S atoms is approximately  $2.41 \text{ \AA}$  and each layer is known as about  $3.15 \text{ \AA}$  thick.(Ataca, Şahin, and Ciraci 2012) The distance information can be used to characterize the layer number of  $\text{MoS}_2$  by using AFM.

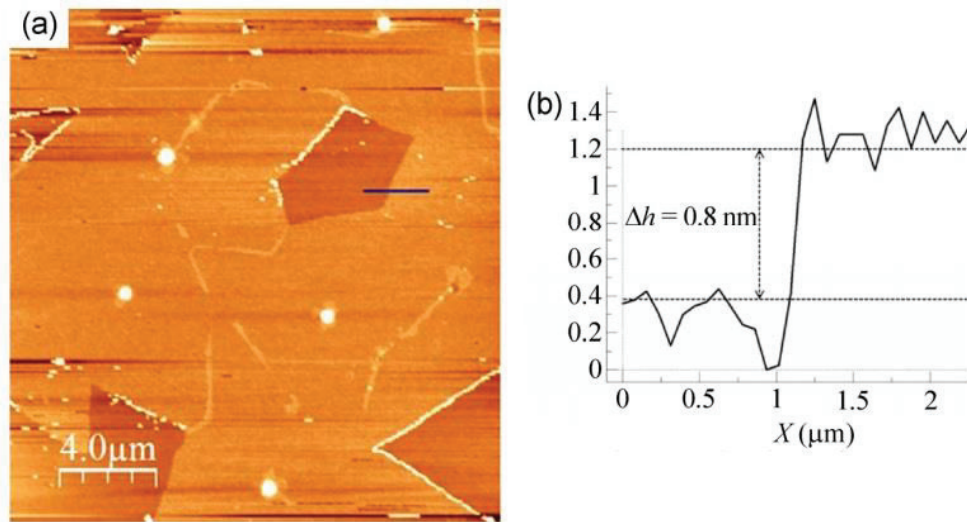


Figure 10. AFM characterization of MoS<sub>2</sub>; a) AFM image of a grown MoS<sub>2</sub> monolayer film, b) height profile of the MoS<sub>2</sub> thin film. (Source : Senthilkumar et al. 2014)

The unique vibrational modes of MoS<sub>2</sub> placed at  $387 \text{ cm}^{-1}$ ,  $408 \text{ cm}^{-1}$  that are  $E_{2g}^1$  and  $A_{1g}$ , respectively. The in-plane vibrational mode is  $E_{2g}^1$ , which is caused by the vibration of Mo and S atoms. The out-plane mode is  $A_{1g}$ , which results from the vibration of S atoms. The thickness of the MoS<sub>2</sub> film affects these two modes, and the frequency distance between modes of the peaks can give information about the film as a monolayer. The frequency difference is around  $19.5 \text{ cm}^{-1}$  for the monolayer MoS<sub>2</sub>. (Splendiani et al. 2010)

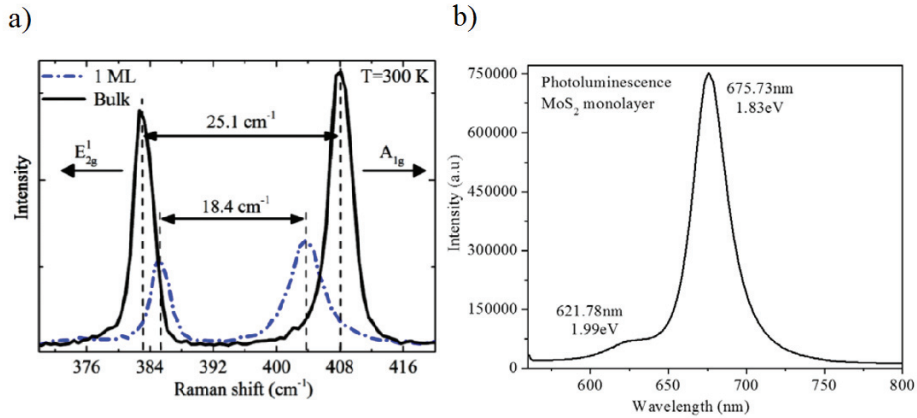


Figure 11. Raman and PL spectra of MoS<sub>2</sub>; a) Raman spectra of different layer number, b) PL spectrum of monolayer MoS<sub>2</sub>.(Source : Splendiani et al. 2010; Ganorkar et al., n.d.)

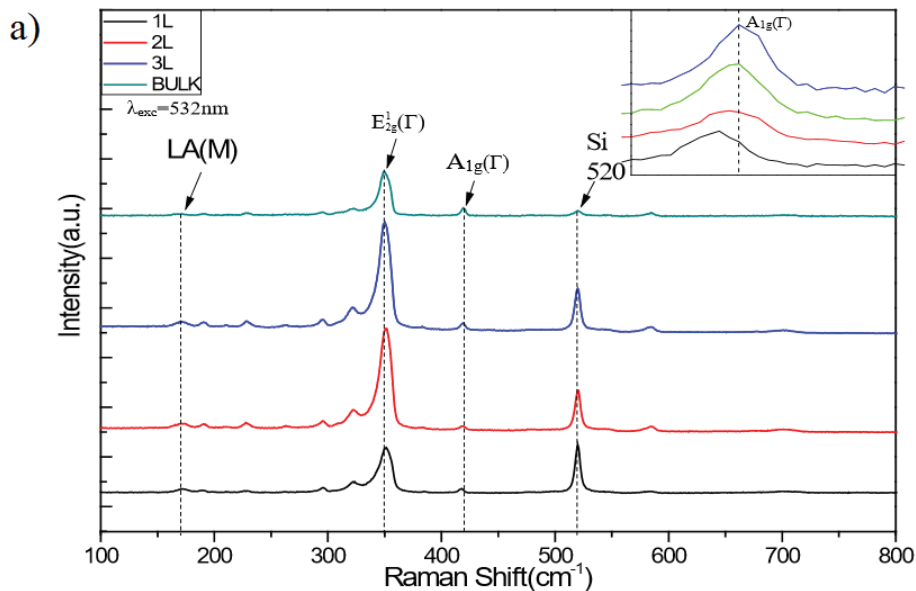
Photoluminescence spectroscopy is a non-destructive, contactless method for characterizing the electronic structure of nanomaterials. The working principle of photoluminescence spectroscopy is based on the photoexcitation principle. Photoexcitation causes electrons to move to the excited state. When the electrons return to the ground state, there can be emitted light or may not. The emitted light is related to the energy difference between the ground state and excited state. The number of layers is closely related to the photoluminescence of MoS<sub>2</sub>.(Mak et al. 2010) The photoluminescence peaks are distinct for different number of layers. Two characteristic absorption peaks of monolayer MoS<sub>2</sub> are placed at 670 nm and 672 nm, which are identified as A1 and B1 exciton peaks.(Splendiani et al. 2010; Ganorkar et al., n.d.) The energy split from the valance band spin-orbit coupling implies the excitons.

### 1.3.5. Tungsten Disulfide

The other similar structured TMDC is a tungsten disulfide (WS<sub>2</sub>). In contrast to graphene, tungsten disulfide offers great promise for electronic and optical devices with layer-dependent tunable electronic bandgap. Also, superior carrier mobility and chemical robustness of WS<sub>2</sub> makes it desirable for a variety of critical applications in electronics and photonics.(Q. H. Wang et al. 2012) A monolayer WS<sub>2</sub> consists of hexagonal lattice structure with metal atom layer, which is sandwiched between sulfur atom layers. While bulk tungsten disulfide has an indirect band gap of 1.3 eV, monolayer WS<sub>2</sub> has a direct band gap of 2.1 eV.(L. Liu et al. 2011) The most obvious

difference of monolayer tungsten disulfide from other TMDCs,  $WS_2$  has a strong photoluminescence emission.(Kang, Godin, and Yang 2015) The important exhibition of  $WS_2$  is strong spin-orbit coupling and band splitting owing to the spin enabling spintronics.(Gao et al. 2015)  $WS_2$  can also be used in many application areas such as sensors, photodetectors, field effect transistors, thanks to its characteristic optical and electrical properties. The characteristic features of a  $WS_2$  are important for the intended application area. The characterization techniques for  $WS_2$  are optical microscopy, Raman spectroscopy, photoluminescence spectroscopy, and atomic force microscopy techniques.

Raman spectroscopy is a powerful technique for identifying the quality and thickness of  $WS_2$ . Tungsten disulfide has two characteristic Raman active modes that are  $E_{2g}^1$  and  $A_{1g}$ . The peak locations differ due to the layer number, while the peaks of bulk  $WS_2$  are at  $348.7\text{ cm}^{-1}$  and  $419.1\text{ cm}^{-1}$ . The peaks of monolayer  $WS_2$  are at  $352.1\text{ cm}^{-1}$  and  $417.5\text{ cm}^{-1}$ .(Qiao et al. 2017) The distance between the two characteristic peaks in the Raman spectrum indicates whether the  $WS_2$  is monolayer or bulk. Furthermore, the layer number of the  $WS_2$  can be estimated by the Raman peak intensities.



b)

	1-layer	2-layers	3-layers	Bulk
$E_{2g}^1(\Gamma)$ ( $\text{cm}^{-1}$ )	352.1	350.9	349.8	348.7
$A_{1g}(\Gamma)$ ( $\text{cm}^{-1}$ )	417.5	418.3	418.7	419.1

Figure 12. a) Raman spectra measurements; Raman shift of different layers of WS<sub>2</sub>

b) The frequency of the Raman shift as a function of the layer number.(Source : Qiao et al. 2017)

Owing to the direct band gap of monolayer tungsten disulfide, photoluminescence emission has been observed in the PL spectrum. Photoluminescence of WS<sub>2</sub> is indeed very strong and hence the PL spectrum can be measured for the characterization of the WS<sub>2</sub>. The only excitonic PL peak is located at  $\sim 620 \text{ nm}$  and, in fact, it proves that WS<sub>2</sub> is monolayer. (Lan et al. 2018)

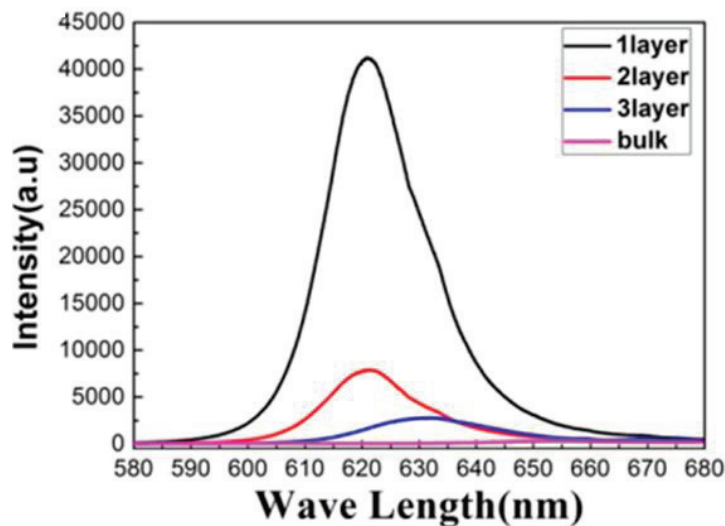


Figure 13. PL spectra of bulk, one-layer, two-layer, and three-layer tungsten disulfide.

The thickness of the crystal can give very important information about layer number, and the thickness can be measured by using atomic force microscopy. The distance between the tungsten and sulfur atoms is around 2.39 Å and the thickness of the WS<sub>2</sub> is around 3.13 Å for each layer. Atomic force microscopy also shows the lateral size and structural defects of the crystal. (Ataca, Şahin, and Ciraci 2012)

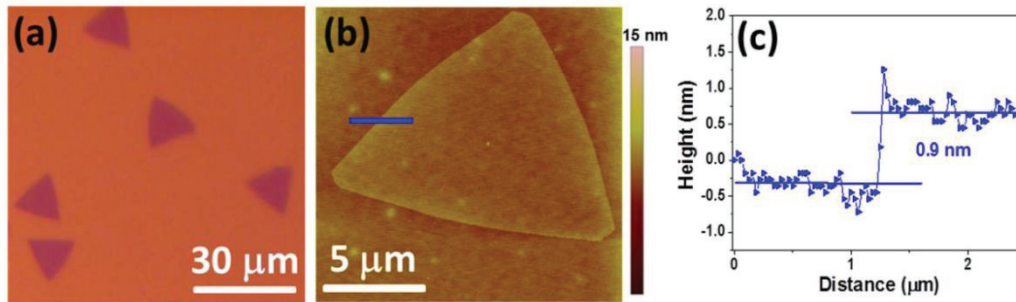


Figure 14. (a) Optical microscopy image, (b) AFM image, and (c) Height profile of WS<sub>2</sub>. (Source : Khairuzzaman 2016)

## CHAPTER 2

### EXPERIMENTAL METHODS AND RESULTS

#### 2.1. Synthesis and Transfer of Graphene

Graphene can be prepared and synthesized by using a variety of techniques such as exfoliation, thermal decomposition, un-zipping CNTs, plasma enhanced chemical vapor deposition, and thermal chemical vapor deposition. The most promising and scalable technique for the growth of graphene is thermal chemical vapor deposition. In this thesis, CVD growth procedure was used as following the previously published graphene synthesis procedure by Salihoglu et al. 2016. The furnace with a 9 in. diameter quartz tube and Cu foils with a thickness of 25 μm were used for the synthesis of single layer graphene on copper foils. After placing Cu foils, the quartz tube was evacuated, because low pressure is needed for the growth process. After evacuation, the furnace was heated to 1035 °C. The partial pressures  $P_{CH_4}$  and  $P_{H_2}$  are 1.5 Torr and 3.5 Torr, respectively. The flow rate of the promoter gases were  $J_{CH_4} = 10$  s.c.c.m and  $J_{H_2} = 85$  s.c.c.m.,

which were used for the full coverage of single layer graphene on copper foils. The growth time was 10 min for the process. After the growth, the samples were cooled down to the room temperature naturally. It should be noted that the ends of the quartz tube was cooled during the annealing step in order to protect plastic O-rings.

The transfer of graphene from copper foil to any other substrate can be done with wet transfer. First, graphene grown on Cu foil was covered by the photoresist with a drop coating method. Photoresist coated samples cured at 65 °C overnight. The Cu foil was etched in an etchant solution of a 1 M iron chloride solution. Subsequently, photoresist/graphene layer was rinsed in deionized water for the cleaning the chemical residue left on graphene. Cleaned photoresist/graphene layer was carefully placed on the target substrate. A hot plate at 75 °C used for the composing of the photoresist/graphene layer to the target substrate for a minute. Lastly, the photoresist layer was washed with acetone, isopropyl alcohol, and deionized water, and then dried with a stream of nitrogen.

## 2.2. Synthesis and Transfer of TMDC

TMDCs can be synthesized with a variety of techniques such as exfoliation, wet chemistry, and chemical vapor deposition. In this thesis, the two-zone CVD method was used for the synthesis of TMDs, molybdenum disulfide and tungsten disulfide. The two methods for the synthesis of MoS<sub>2</sub> and WS<sub>2</sub> have a very similar process. They differ with the powder amount, and also temperature depending on the type of the powder. The optimization of the process for synthesizing MoS<sub>2</sub> and WS<sub>2</sub> was done by following the published work by F. Zhang et al. 2019.

Table 1. CVD synthesis parameters of MoS<sub>2</sub> and WS<sub>2</sub>.

	Temperature	Ar Flow	MoO <sub>3</sub> /WO <sub>3</sub>	S
MoS <sub>2</sub>	780	100 s.c.c.m.	15 mg	200 mg
WS <sub>2</sub>	825	100 s.c.c.m.	10 mg	200 mg

## 2.3. Synthesis and Transfer of Molybdenum Disulfide

A powder-based atmospheric pressure CVD method in an inert atmosphere was used for the synthesis of MoS<sub>2</sub>. One of the quartz boats with a powder mixture of 15 mg (the ratio 5:1) MoO<sub>3</sub> and NaBr was carefully placed at the center of the furnace, and then the other quartz boat with a 200 mg sulfur powder was placed in the furnace at 200 °C. A 300 nm Si/SiO<sub>2</sub> substrate was placed face down on the quartz boat that has the Mo source. The SiO<sub>2</sub> substrate was carefully cleaned in a Piranha solution, which is a mixture of concentrated sulphuric acid with hydrogen peroxide in a ratio of 3:1 to 6:1 (*Caution: piranha solution reacts very violently with organic matters.*). Firstly, the furnace was flushed with excess amount of Ar gas, 1000 s.c.c.m Ar, for 15 minutes. The furnace was heated to 780 °C in 30 min. During the whole growth process, 100 s.c.c.m. Ar flow was used as a carrier gas. After the furnace reaches to 780 °C, around 15 minutes is enough for the deposition of the MoS<sub>2</sub> flakes. Finally, the furnace was opened completely for finishing the growth of MoS<sub>2</sub>, and Ar flow was continued to cool down the hot furnace.

As also commonly used in the literature, a PMMA assisted wet transfer method was used for the transfer of MoS<sub>2</sub> to the target substrate. First, a PMMA layer was spin-coated on the as grown MoS<sub>2</sub> for 2 minutes at 2000 rpm, and then it was baked at 130 °C for 3 minutes. A PMMA covered MoS<sub>2</sub>/Si/SiO<sub>2</sub> substrate was immersed in 2 M NaOH solution on hot plate at 100 °C. After the etching of SiO<sub>2</sub>, PMMA/MoS<sub>2</sub> layer was put into DI water with a fishing out technique. The DI water was used for cleaning any residue of the etcher solution. Subsequently, PMMA/MoS<sub>2</sub> layer was placed on the target substrate and it was baked for around 5 min at 200 °C for drying. Finally, the PMMA layer was washed with acetone and isopropyl alcohol.

## **2.4. Synthesis and Transfer of Tungsten Disulfide**

A similar process used for the synthesis of MoS<sub>2</sub> was used for the synthesis of tungsten disulfide, WS<sub>2</sub>. The main differences were the temperature of the annealing furnace and the powder amount used for the deposition of metal and Sulphur sources. The furnace was heated to the 825 °C in 30 minutes. The powder amounts were 10 mg WO<sub>3</sub> and NaBr with the ratio 5:1 and 200 mg sulfur. The Sulfur needs the same



temperature of 200 °C for the growth. The transfer of WS<sub>2</sub> is very similar to the transfer process used for MoS<sub>2</sub>, as can be read above.

## **2.5. Synthesis and Purification of Silver Nanowire**

All chemicals were used without any further purification as they were obtained. The PVP was dissolved in a solution of 10 mL glycerol and an amount of EG for a standard synthesis. Meanwhile, in 5 mL of glycerol solution, 0.034 g of AgNO<sub>3</sub> was dissolved. The glycerol and AgNO<sub>3</sub> solution added to flask for the heating in oil bath. Then the oil bath heated to 170 °C in 30 minutes with stirring the solution. An amount of catalysts (CuCl<sub>3</sub>, CoCl<sub>3</sub>, MnCl<sub>3</sub>, and CrCl<sub>3</sub>) solution was added drop wisely to the solution. At the end of the two hours, the synthesized AgNW solution was cooled down fastly with an ice bath.

For the purification of AgNWs, the solution was diluted with 30 mL of deionized water. Acetone (80-160mL) was dripped into it by stirring slowly, while the nanowires were clustered. It was waited for 10 minutes for the particles to settle to the bottom of the bottle and the supernatant on top was replaced with a pipette. 0.5% PVP was added into 20 mL of distilled water, the clustered nanowires were re-dissolved in this water, 40-80 ml of acetone was added to precipitate and this procedure was repeated 4 times to obtain high purity nanowires. Finally, the nanowires were dispersed in isopropyl alcohol, and then the nanowires were centrifuged and recovered from the excess PVP.

## **2.6. Optical Measurements**

Optical transmission of AgNW thin films and the quality (length, diameter, and distribution on the surface after coating) of AgNWs were tested by using a spectrophotometer, an optical microscope, and a scanning electron microscope. The nanowire length and diameter directly affect the transmission performance of the electrode fabricated from silver nanowires. Thinner and longer nanowires were expected for the maximization of the transmission of the nanowire electrode. The transmission of the electrodes is around ~87% and the graphene effect on the transmission is ~2.5%. Optical properties of the electrodes will be discussed in the result and discussion section.

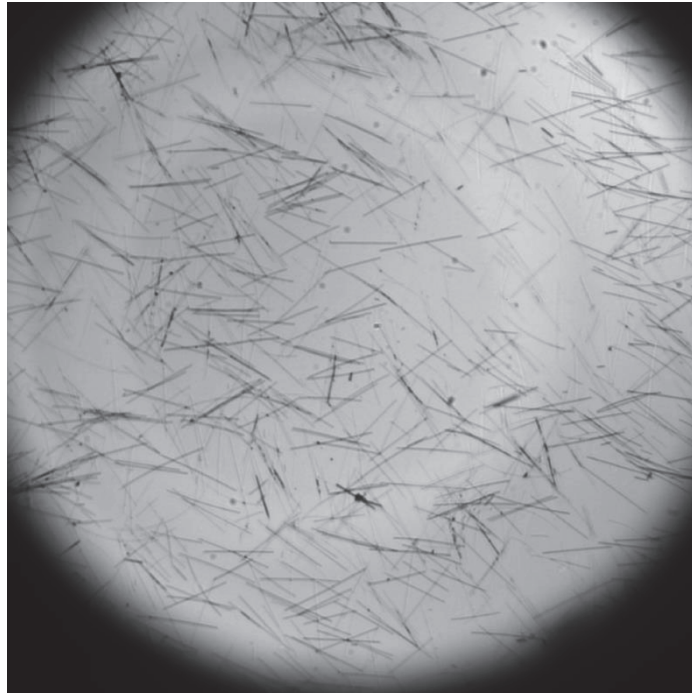


Figure 15. Optical microscopy image of Co catalyzed AgNWs on a glass substrate.

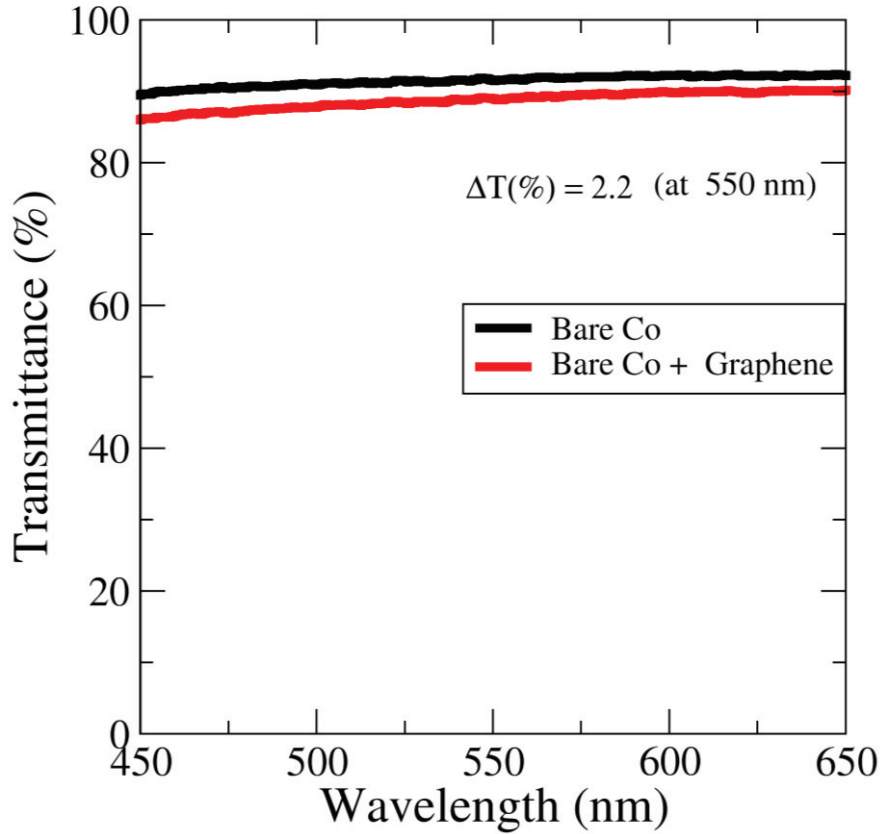


Figure 16. Transmittance of the Co catalyzed electrode containing Ag nanowires on a glass substrate.

The black curve indicates transmission of the transparent conducting electrode fabricated from Co catalysed Ag NWs. The red curve shows transmission of the graphene coated transparent electrode fabricated from Co catalysed Ag NWs. The decrease in transmission in the graphene coated transparent electrode is due to the interband transitions in graphene in the visible region of the electromagnetic spectrum.

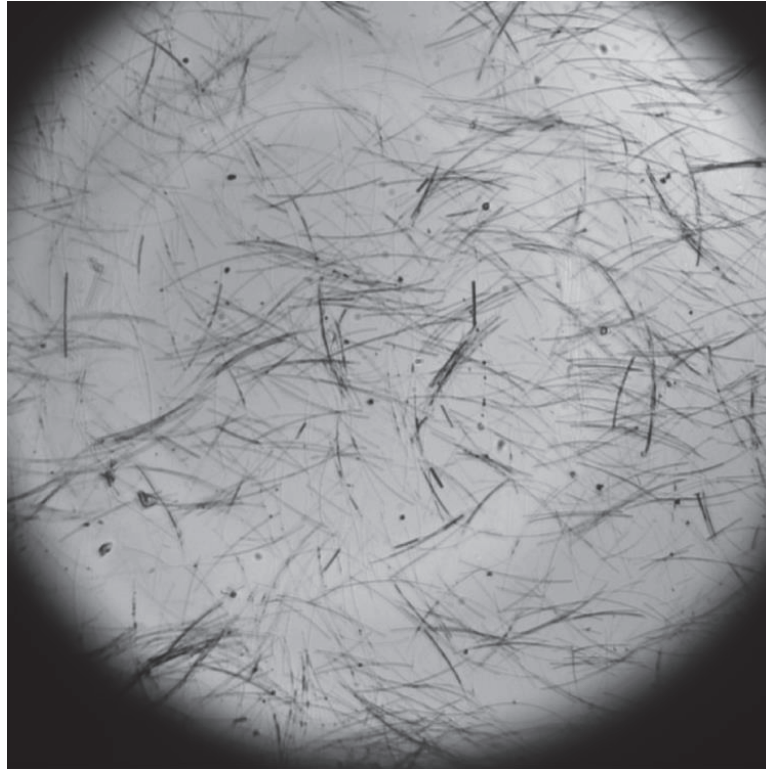


Figure 17. Optical microscopy image of Cr catalyzed AgNWs coated on a glass substrate.

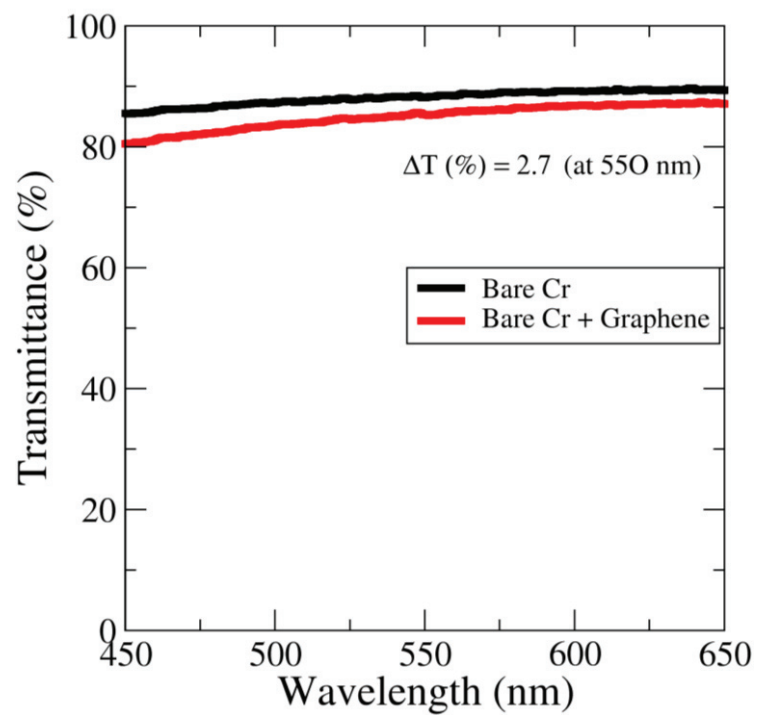


Figure 18. Transmittance of the Cr catalyzed electrode containing Ag nanowires on a glass substrate.

The black curve indicates transmission of the transparent conducting electrode fabricated from Cr catalysed Ag NWs. The red curve shows transmission of the graphene coated transparent electrode fabricated from Cr catalysed Ag NWs. The decrease in transmission in the graphene coated transparent electrode is due to the interband transitions in graphene in the visible region of the electromagnetic spectrum.

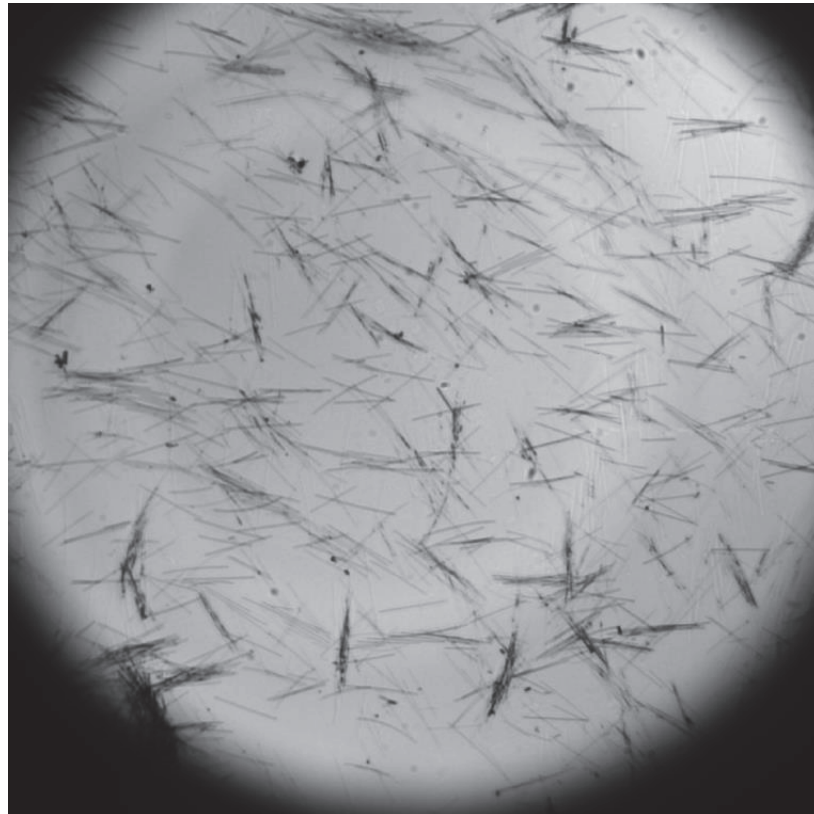


Figure 19. Optical microscopy image of Mn catalyzed AgNWs on a glass substrate.

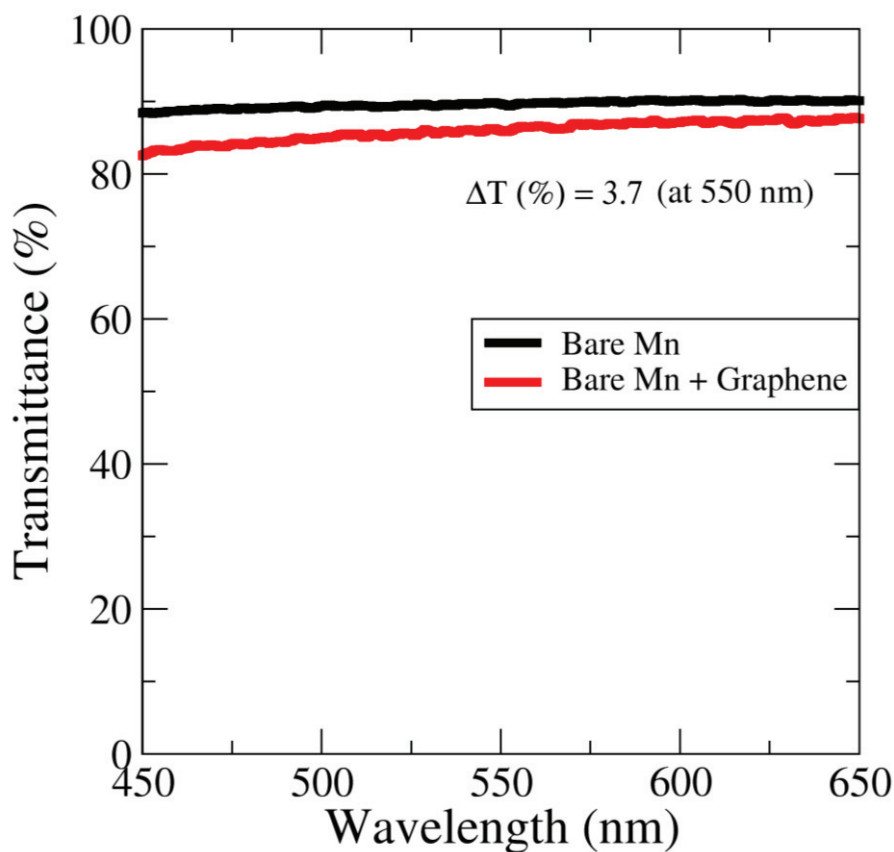


Figure 20. Transmittance of the Mn catalyzed electrode containing Ag nanowires on a glass substrate.

The black curve indicates transmission of the transparent conducting electrode fabricated from Mn catalysed Ag NWs. The red curve shows transmission of the graphene coated transparent electrode fabricated from Mn catalysed Ag NWs. The decrease in transmission in the graphene coated transparent electrode is due to the interband transitions in graphene in the visible region of the electromagnetic spectrum.

The black curve indicates transmission of the transparent conducting electrode fabricated from Cu catalysed Ag NWs. The blue curve shows transmission of the graphene coated transparent electrode fabricated from Cu catalysed Ag NWs. The red curve represents transmission of graphene and WS<sub>2</sub> coated electrode fabricated from Cu catalysed Ag NWs.

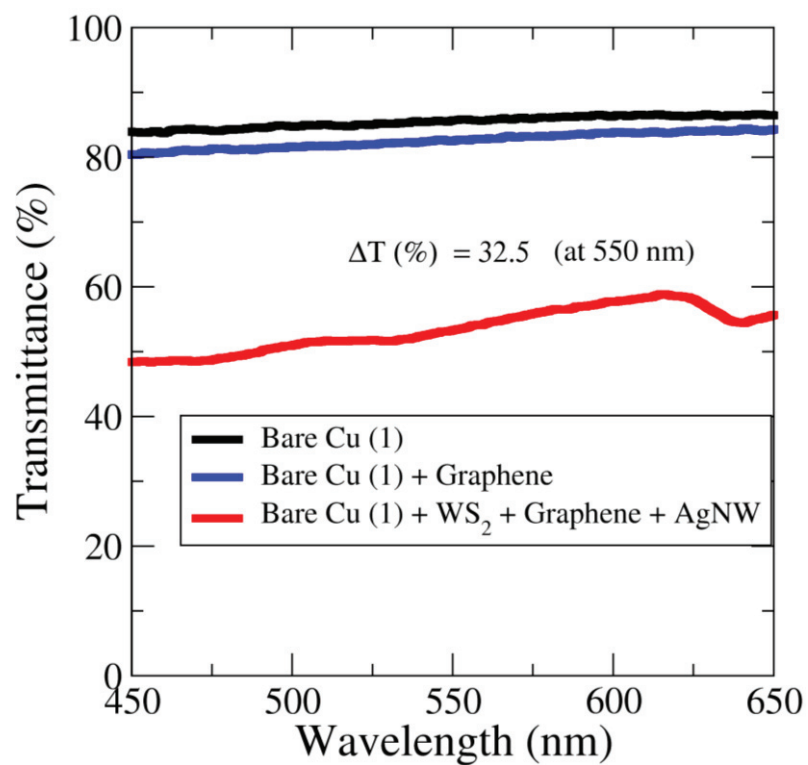


Figure 21. Transmittance of the Cu catalyzed electrode containing Ag nanowires, graphene and WS<sub>2</sub> on a glass substrate.

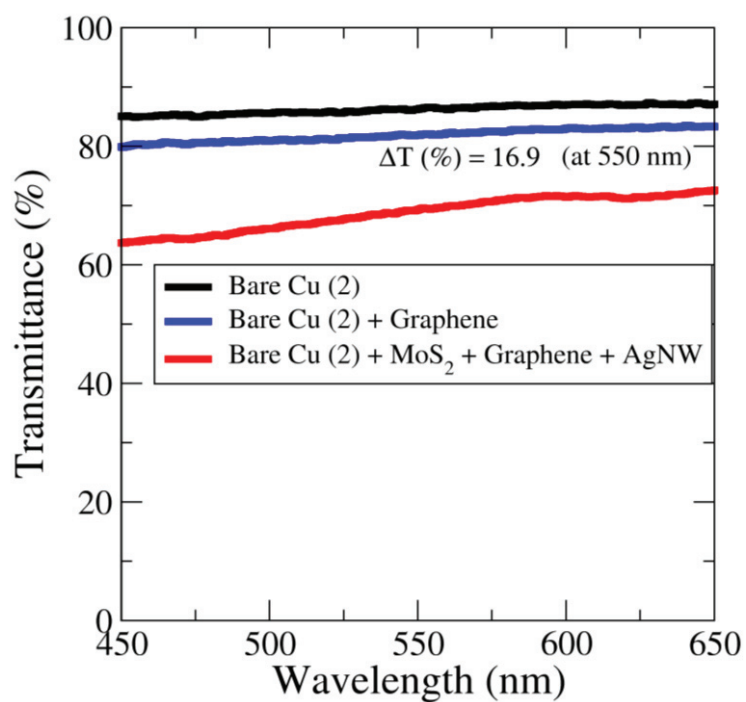


Figure 22. Transmittance of the Cu catalyzed electrode containing Ag nanowires, graphene and MoS<sub>2</sub> on a glass substrate.

The black curve indicates transmission of the transparent conducting electrode fabricated from Cu catalysed Ag NWs. The blue curve shows transmission of the graphene coated transparent electrode fabricated from Cu catalysed Ag NWs. The red curve represents transmission of graphene and MoS<sub>2</sub> coated electrode fabricated from Cu catalysed Ag NWs.

## **2.7. Electrical Measurements**

The most important factor determining the electrical quality of the transparent conducting electrode is the sheet resistance of the electrode, a measure of resistance of conducting thin films. In this thesis, AgNW, graphene, and monolayer TMDC based transparent conducting electrodes were fabricated. The effect of variety of catalysts (Co, Cu, Mn, and Cr) in the synthesis of AgNWs was extensively studied for finding the best AgNWs for the transparent conducting electrodes. The length and diameter of the AgNWs are key parameters for determining the quality of the nanowires in the electrode fabrication. The sheet resistance of the electrodes fabricated from the AgNWs can give information about the electrical properties of the fabricated transparent conducting electrodes. The lowest resistance of the transparent conducting electrodes was expected for the highest conductivity of the electrodes. A multimeter was used to measure the resistance of the electrodes. The effect of annealing temperature on the resistance of the transparent conducting electrode will be discussed in the result and discussion section.



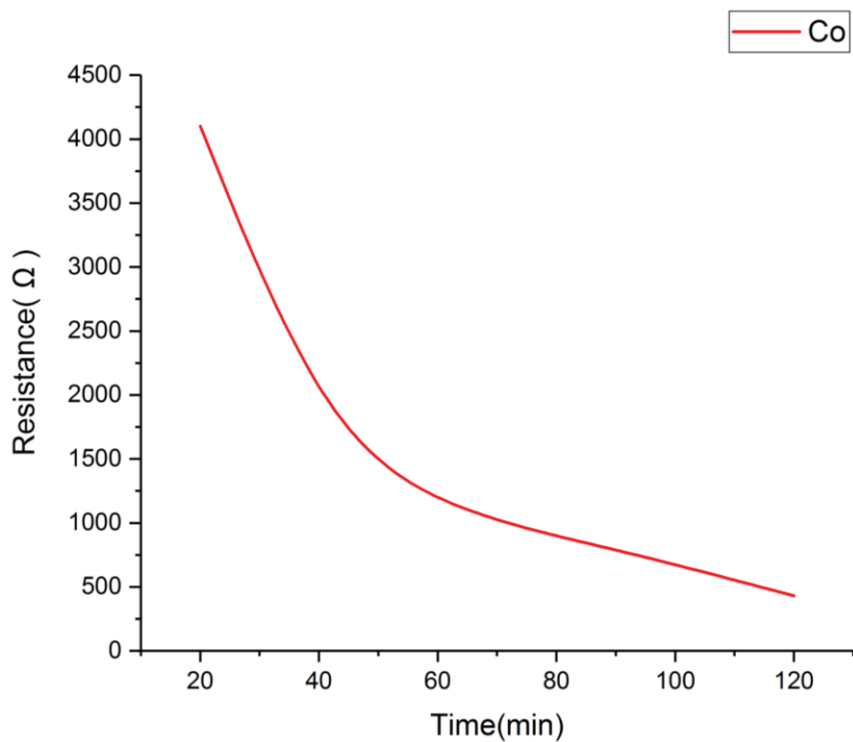


Figure 23. Resistance change of transparent conducting thin films fabricated by using Co catalyzed AgNWs with annealing time.

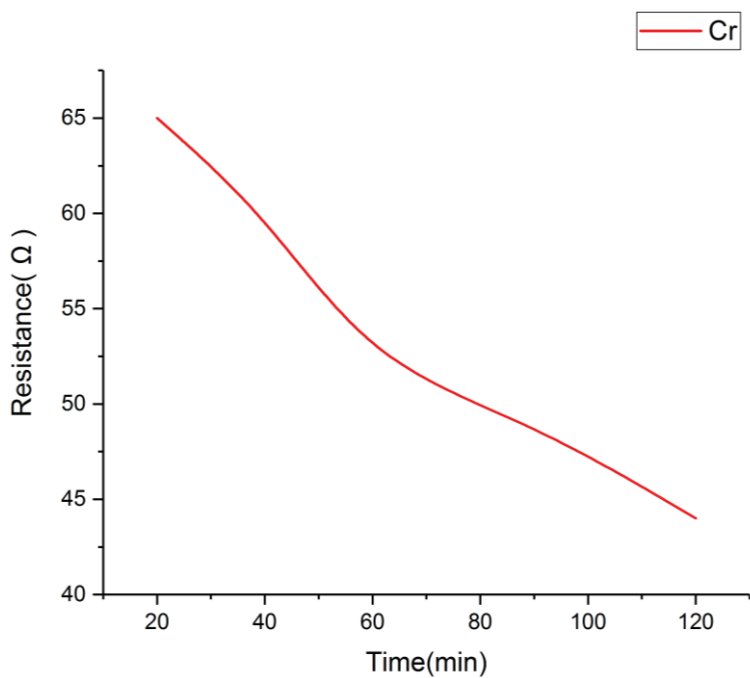


Figure 24. Resistance change of transparent conducting thin films fabricated by using Cr catalyzed AgNWs with annealing time.

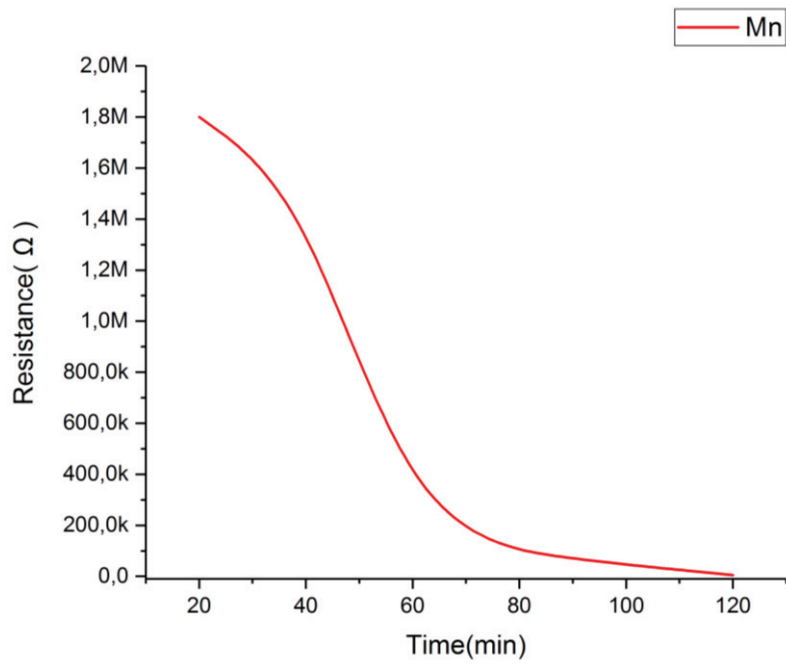


Figure 25. Resistance change of transparent conducting thin films fabricated by using Mn catalyzed AgNWs with annealing time.

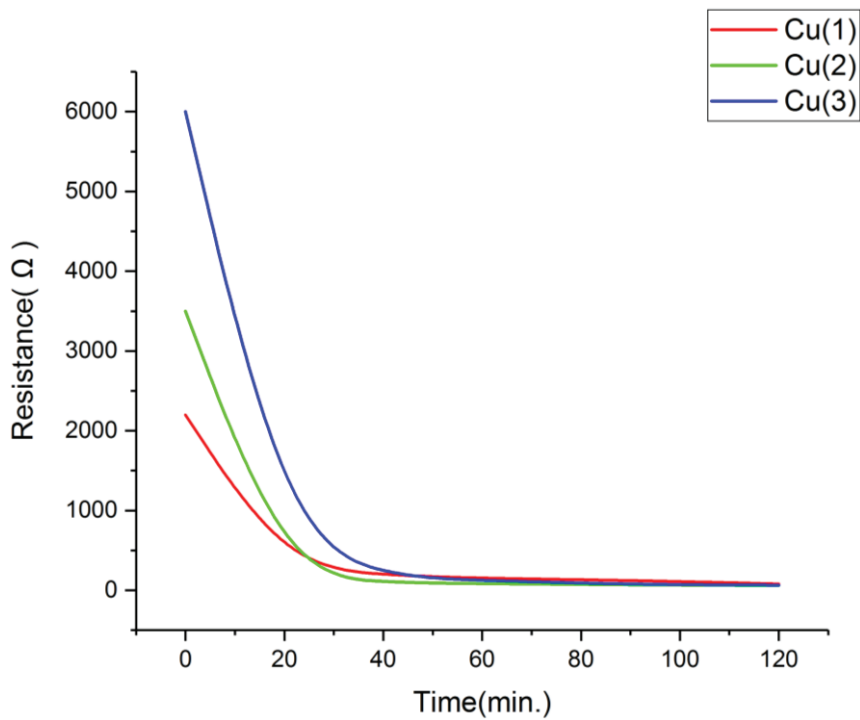


Figure 26. Resistance change of transparent conducting thin films fabricated by using Cu catalyzed AgNWs with annealing time.

The flexible and transparent conducting electrodes were fabricated on flexible substrates. The flexibility of electrodes was tested by the 90° bending test. The bending was done 90 times for four different flexible electrodes. The electrodes were fabricated from Ag NWs synthesized by using four different catalysts, namely CuCl<sub>3</sub>, CoCl<sub>3</sub>, MnCl<sub>3</sub>, and CrCl<sub>3</sub>. The bending effect is directly related to the length of the nanowires. The effect of bending will be discussed in the result and discussion section. Below in Figures 27-30, we provide the results obtained from the bending tests.

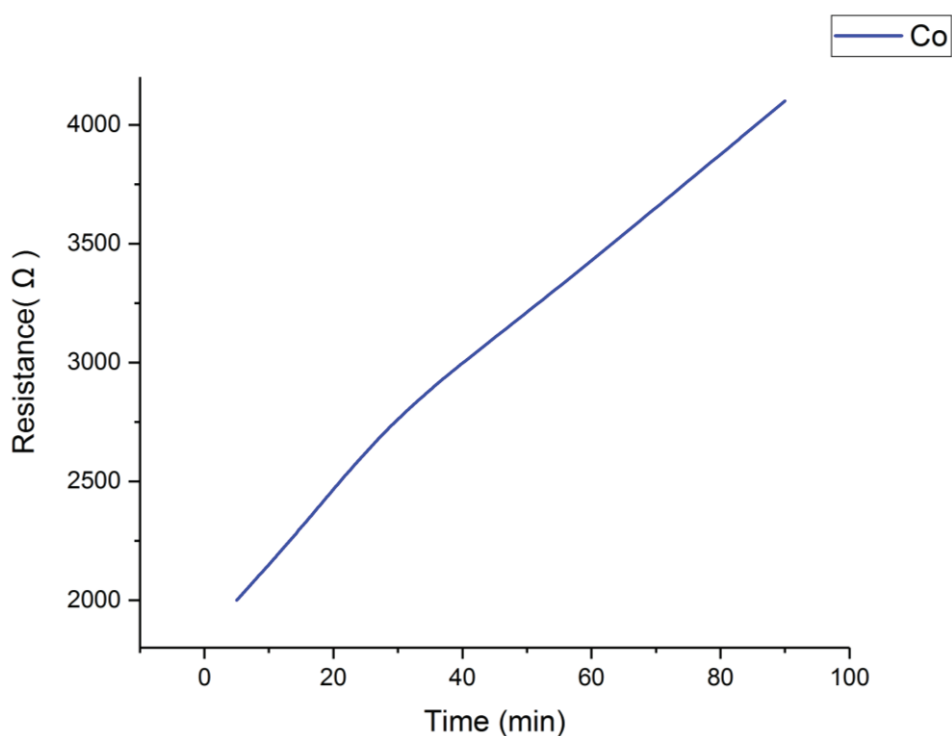


Figure 27. Bending effects on the resistance of the transparent conducting electrode fabricated by using Co catalyzed AgNWs.

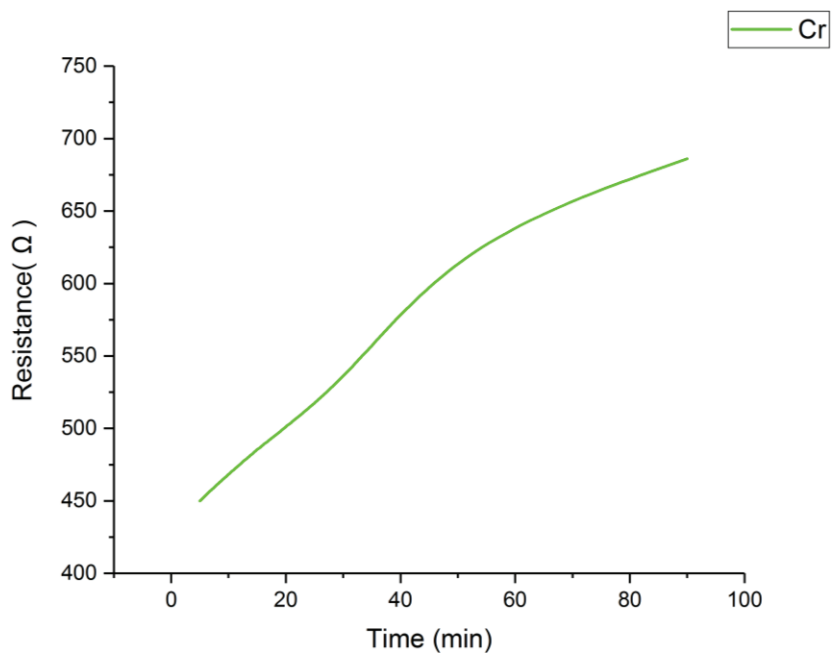


Figure 28. Bending effects on the resistance of the transparent conducting electrode fabricated by using Cr catalyzed AgNWs.

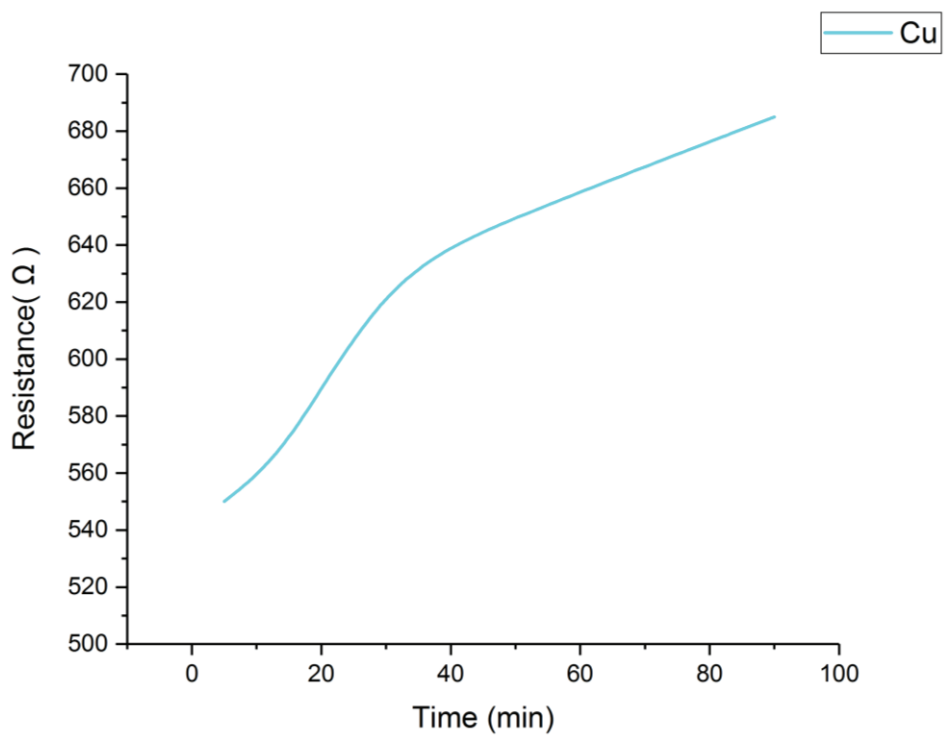


Figure 29. Bending effects on the resistance of the transparent conducting electrode fabricated by using Cu catalyzed AgNWs.

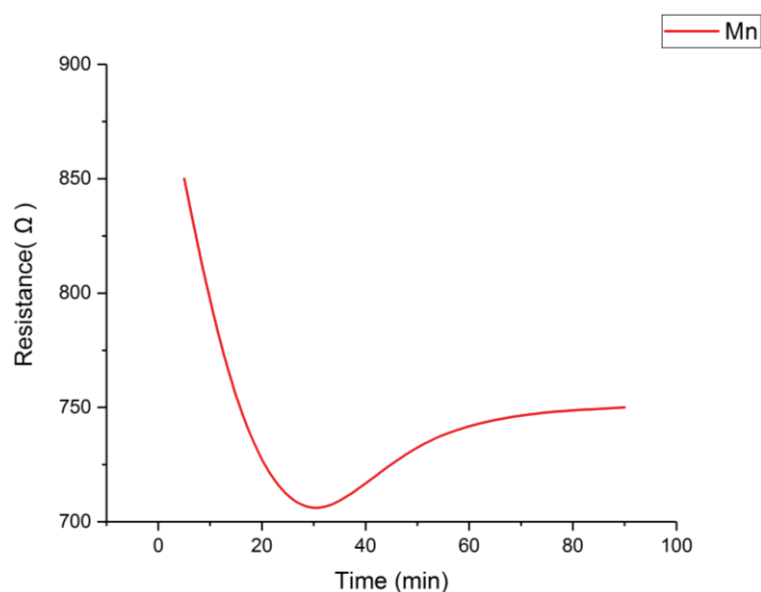


Figure 30. Bending effects on the resistance of the transparent conducting electrode fabricated by using Mn catalyzed AgNWs.

Table 2. Resistance variation of graphene-AgNWs hybrid conductive electrodes.

	Resistance ( Ω )			
	Cu	Mn	Cr	Co
<b>AgNW+Graphene</b>	210	830	600	900
<b>After Anneal(2h)</b>	80	200	65	170
<b>Flexible Form</b>	500	1000	950	900

The Figures 23-30 show the effect of Cu, Mn, Cr, and Co catalysts used during the synthesis of AgNWs on the nanowire length and diameter, and electrical properties of the fabricated transparent conducting thin films. Table 2 summarizes the results obtained from these experiments.

## CHAPTER 3

### RESULTS AND DISCUSSIONS

The optical and electrical properties of the conductive electrodes are directly related to the structure of the AgNWs deposited on the solid and flexible substrates. Optical microscopy images of the AgNWs on glass substrates clearly show the length effect on the transparency of the conductive electrodes. Comparing the Cr and Mn catalyzed AgNWs, Cr catalyzed AgNWs are longer than the Mn and Co catalyzed AgNWs. This can be shown by the resistance value of the electrode before annealing; Cr electrode has 65 ohm resistance at the beginning of the annealing but the Mn electrode has 1.8 Mohm and Co electrode has 4 kohm resistances. The resistance data can be useful for the estimation of the nanowire length. However, scanning electron microscopy investigations of the AgNWs give us detailed information about the length and diameter of the nanowires. Annealing process is essential for the purification and welding of the AgNWs. Therefore, significant resistance decrease of the AgNWs has been observed by the annealing. As a result of annealing, the resistance tends to decrease to close value, even with the same type of AgNW with different transmission, Fig. 25. The polymer residues on the AgNWs were completely removed and the nanowires contact points were welded by the thermal annealing. The effect of annealing could not be observed on the transmission measurements. The 2.3% transmission decrease is observed after the graphene transfer to the conductive electrodes. (Salihoglu, Kakenov, et al. 2018) As shown in the transmittance figures, ~2.5% transmittance decrease has been observed. Owing to the interband transitions in graphene, graphene decreases transmission in the visible region of the electromagnetic spectrum about 2.3%. The difference between the theoretical and experimental data is due to the chemical residues present on the graphene coming from the graphene transfer process. Moreover, graphene transfer has dramatically improved the stability of AgNWs as a protective conductive layer. The thick layers of TMDC causes the downfall of the transmission, Fig 20 and Fig 21. Further experiments are definitely needed in order to find the optimum thickness of the TMDCs by looking at the resistance and transmission of the thin conducting film.

Flexibility test of the conductive electrode was performed by the bending process. All of the AgNWs synthesized by using different catalysts were carefully tested. The resistance value increased after bending the AgNWs but the change in resistance is negligible. There is not any considerable increase at the resistance value. Only Mn catalyzed AgNW electrode showed different characteristic at the bending test. All of these results show that AgNWs together with single layer graphene are promising candidate for next generation flexible transparent conductive electrodes. However, further experiments are definitely needed in order to fabricate graphene-AgNWs-TMDs transparent, conducting, and flexible electrodes. These electrodes can find applications in flexible electronics, solar cells, light emitting diodes, optical modulators, displays, touch and pressure sensors as well.

## CHAPTER 4

### CONCLUSION

After the discovery of graphene, the development of 2-dimensional materials has opened up new opportunities for applications in optoelectronic devices. Efforts to reduce the size of materials in the development of new technologies have led to the requirement for new types of electronic devices. Graphene, AgNWs, and TMDCs allowed us to increase performance of optoelectronic devices. In this thesis, flexible and transparent conductive electrodes based on AgNWs, graphene, and TMDCs have been developed and the composite structured electrode has been extensively studied. The main results achieved in this thesis can be summarized below. 1-) Large area single layer graphene was synthesized on copper foils in a CVD furnace and transferred to any rigid and flexible substrates. 2-) Silver nanowires with very small diameter and very long lengths were synthesized at high temperature in an oil bath. The mechanism of the silver nanowire formation and effects of catalysts on nanowire length and diameter have been studied extensively. 3-) Single and multi layer MoS<sub>2</sub> and WS<sub>2</sub> flakes were synthesized in a CVD furnace in an inert atmosphere. The single layer TMDCs synthesized in this work will find applications in optoelectronic device applications in future studies. 4-) Flexible and transparent conducting electrodes on rigid and flexible substrates fabricated by using single layer graphene, silver nanowires, and multilayer TMDCs. This composite structure (AgNW, graphene, and TMDC) is promising for the future applications. Further experiments are definitely required fully integrate single layer graphene with silver nanowires and a few layers TMDCs.



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