Solubility and aging of lead magnesium niobate in water

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Abstract

Lead magnesium niobate (PMN) is an important relaxor ferroelectric material commonly employed in multilayer capacitor and actuator manufacturing owing to its high dielectric constant and superior electrostrictive properties. However, stability of this material in water is not very well known and there is need for a detailed investigation. In this research, solubility of lead magnesium niobate powders in water was determined as a function of solids concentration. The obtained results showed that the amount of cation leaching from the PMN surface depends on the pH value of the suspension and the solids concentration. The Pb2+ and Mg2+ ion dissolution was very high especially in the acidic pH range. Nevertheless, neither the dissolution mechanism nor the effects of dissolved ions on the stability were the same for those ions. The study provides new aspects on the solubility of perovskite materials which possess more than one soluble cation in their structure.

Keywords: A. Suspensions; B. Surfaces; D. Perovskites; Ceramics; Relaxor ferroelectrics

1. Introduction

Lead magnesium niobate (PMN) is a relaxor ferroelectric material which is characterized by a diffuse phase transition over a broad temperature range and a frequency dependent maximum in its relative dielectric permittivity [1,2]. It has many potential applications such as actuators, motors, pumps, optical scanning systems, and vibration isolators [3,4]. The most common application of PMN is in multilayer ceramic capacitors (MLCC).

The manufacturing of electronic devices such as MLCC composed of perovskite materials generally involves a colloidal process which requires preparation of colloidal suspensions. The previous studies on this subject include the preparation of lead magnesium niobate–lead titanate (PMN–PT) suspensions in solvents other than water [5–7]. PMN–PT powders can be dispersed in organic solvents, to get well-dispersed suspensions. However, the use of aqueous media instead of organic solvents is desirable nowadays due to economic and environmental considerations [8]. Therefore, understanding the PMN–H2O system is essential in producing high quality PMN ceramics. On the other hand, the behavior of PMN particles in water medium has not been widely reported.

Most perovskites are not thermodynamically stable in water. Metal ions in the ceramic powders are dissolved in water, which affects the dispersion stability of the suspension [8,9]. Perovskite materials such as PMN have the general chemical formula, ABO3 (A = Na, Ca, Ba, Pb, Mg, etc.) and (B = Ti, Nb, Zr, etc.) [10]. In perovskites the cation at the A site is generally soluble in water whereas the B site cation that is only slightly soluble overall the practical pH range. Therefore, A site cation undergoes incongruent dissolution leaving an interface which is relatively rich in the B site cation [11].

In this study solubility characteristics of the cations that constitute PMN was examined as a function of suspension pH and solids concentration. Effect of cation leaching on the surface bond formation and the microstructure of the PMN powders was also investigated.

2. Experimental

Lead magnesium niobate, Pb(Mg1/3Nb2/3)O3 powder which was produced by combustion spray pyrolysis method provided by Praxair Specialty Ceramics (Woodinville WA). Powder purity is 99.9% as reported by the manufacturer. Bulk density
(Helium pycnometer Micromeritics 1330), and the BET surface area (Micromeritics ASAP 2400, Norcross, GA) of the powder are measured to be 7.967 g/cm$^3$ and 1.168 m$^2$/g, respectively. Particle size distribution of the powder was obtained using a particle size analyzer (Model CAPA-700, Horiba, Ltd., Tokyo, Japan) and the d$_{50}$ of the powder was measured to be 2.03 μm. Crystal structure of the PMN powder was identified using X-ray diffractometer (Philips, XL30-S FEG). Morphology was investigated using Scanning Electron Microscope (Philips X’Pert Pro). Fig. 1 shows the microstructure and the crystal phase of the PMN powder used in the study.

Aqueous PMN suspensions were prepared at different solids loadings ranging from 0.1 to 10 vol% using deionized water. Suspensions were magnetically stirred for 24 h prior to the measurements at room temperature. Adjustments of the pH were performed using 1 M solutions of HNO$_3$ and NH$_3$OH. The pH values of the suspensions were controlled during the stirring period and further adjustments were performed when necessary.

Solubility of Pb$^{2+}$, Mg$^{2+}$ and Nb$^{5+}$ ions in the PMN structure was examined using Inductively Coupled Plasma Mass Spectroscopy (ICP-MS Agilent, 7500 Series) and Inductively Coupled Plasma Atomic Emission Spectroscopy (ICP-AES Optima 2000 DV PerkinElmer Norwalk, CT, USA). For this purpose suspensions prepared at different solids loadings according to the procedure described in the previous section were centrifuged at 4000 rpm for 1 h to obtain a clear supernatant. The supernatants were further filtered from 0.02 μm microfilters to eliminate any possible particles in solution. The amounts of Pb$^{2+}$, Mg$^{2+}$ and Nb$^{5+}$ in particle-free supernatants were determined using ICP-AES (for 5 vol% suspensions) and ICP-MS (for other solid concentrations).

Mobility of PMN suspensions were directly determined by electrokinetic sonic amplitude measurements using an electro-acoustic analyzer (ESA-9800, Matec Applied Sciences, Northborough, MA). To ensure dispersion, the suspensions were ultrasonicated for 2.5 min using an ultrasonic horn (550 Sonic Dismembrator, Fisher Scientific, Pardubice, Czech Republic) prior to measurements. The ESA measurements were checked against a 10 vol% Ludox suspension having a zeta potential of $-38$ mV. Experiments were performed triplicate, and the results given are the average of 10 measurements.

Structural bonds of the PMN powders after aging in water at different conditions were analyzed using Fourier Transform Infrared Spectroscopy (FTIR-8201, Shimadzu, Japan). Microstructure and crystal structure of the PMN powders aged in water at different conditions prior to a heat treatment at 1100 °C for 2 h (heating rate, 5 °C/min) was examined using a scanning electron microscope (Philips, XL30-S FEG) and X-ray diffractometer (Philips, X’Pert Pro), respectively.

### 3. Results and discussion

In the first part of the study cation dissolution from the PMN surface was examined as a function of solution pH. The amount of cation concentration in solution obtained from a suspension having a solids concentration of 5 vol% was investigated in our previous study [14]. Accordingly, Fig. 2 shows that the lead and magnesium ion concentrations increase with decreasing pH. Especially the Pb$^{2+}$ concentration increases dramatically in the acidic region. Though the Mg$^{2+}$ dissolution was also pH dependent the dissolution rate was almost an order of magnitude smaller compared to that observed with Pb$^{2+}$ at low pH values. But it was observed that Mg$^{2+}$ dissolution is more significant at neutral pH values. On the other hand, Nb$^{5+}$ exhibited rather low dissolution under the same conditions. The Nb$^{5+}$ concentrations in the acidic region were below the detection limit of the analyzing device [14].

Previously Neubrand et al. [12] investigated the dissolution of Ba$^{2+}$ ions from the barium titanate in water. They found that the amount of barium leached increases as the pH decreases. In fact the increase in the dissolved metal ion concentration with decreasing pH considered to be a general behavior for most perovskites in water [13]. However, in Pb–Mg–Nb–H$_2$O system A-site cations, Pb$^{2+}$ and Mg$^{2+}$, are soluble that makes the surface chemistry more complex compared to other perovskites such as BaTiO$_3$ and PZT.
Adair et al. [11] reported the dissolution types dictated by the ceramic materials and solution pH. According to their work, the most multi-component metal oxides such as barium titanate generally follow incongruent type dissolution and the dissolution mechanism strongly depends on the pH value.

In the current study, cation dissolution from PMN surface was investigated at some other solid concentrations namely 0.1, 1 and 10 vol%. The results are given in Figs. 3–5. According to Figs. 3 and 4, Pb$^{2+}$ ion concentration increases dramatically starting from pH 6 for low solids loading suspensions. However, the increase in cation concentration for higher solids loading suspensions was rather low. The dissolution percentage of Pb$^{2+}$ and Mg$^{2+}$ as a function of solids concentrations is drawn in Fig. 5, showing that cation dissolution from PMN surface increases as the solids concentration decreases.

The increase of cation dissolution at low solids loading suspensions can be explained based on the transport process of particles in the liquid phase. When the PMN is suspended in water a surface layer is developed due to the deficiency of the dissolving Pb$^{2+}$ and Mg$^{2+}$ ions. Specifically, the surface layer forms because of the mass transport limited dissolution. Without the mass transport limitation the entire particle would be converted to the niobium oxide at low solids concentration and in acidic pH conditions. This means that, the surface layer on PMN particles in acidic region will be thicker in dilute suspensions compared to the concentrated suspensions. As much as the rate of surface layer growth in concentrated suspensions is inhibited by the mass transfer, the dissolution rate will be lower in concentrated PMN suspensions [14].

Previously Paik and Hackley [15] investigated the dissolution of barium titanate in water as a function of solids concentration. Accordingly, in barium titanate/water system, Ba$^{2+}$ ions dissolve from the surface, as a function of solids concentration. Therefore, it is possible to say that the results of the current study are in good agreement with the previous study of Paik and co-workers. Fig. 6 displays the kinetics of the Pb$^{2+}$ and Mg$^{2+}$ dissolution at pH values 6 and 9. Depending on this, the dissolution rate of both cations was nearly constant at pH 9 within the time period studied. Nonetheless, dissolution rates were quite different at pH 6. At this pH the lead concentration reaches considerably high levels immediately after the initial contact of powder with water and decreases gradually with time as seen from Fig. 6. The decrease in the Pb$^{2+}$ ion concentration in solution phase may be attributed to the re-adsorption of the
free Pb$^{2+}$ ions onto the hydrated PMN surface. As it is described above, dissolution of Pb$^{2+}$ and Mg$^{2+}$ ions were high at this pH value compared to the dissolution rate at pH 9. On the other hand, Mg$^{2+}$ ion concentration showed gradual increase with time at pH 6 and reached a maximum value after 24 h, indicating a decrease for longer contact times. According to this picture it is possible to conclude that Mg$^{2+}$ and Pb$^{2+}$ ions demonstrate different behavior after they dissolve from the PMN surface. Mg$^{2+}$ and Pb$^{2+}$ ions possibly form complexes with the –OH ions in water. The increase in the pH of the PMN suspensions from pH 6 to 8.6 may support this argument (see Fig. 7). This increase in pH may be attributed to the complex formation between the cations (notably Pb$^{2+}$) dissolved from PMN surface and the –OH ions in water [16]. Figs. 8 and 9 denote the formation pH of various lead and magnesium hydroxides. According to these figures, formation of PbOH or Pb$_4$(OH)$_8$$^{4+}$ may be responsible for the rapid increase in pH of the PMN suspension (pH$_{initial}$ 6). Fig. 7 also depicts that the suspensions most probably consume the protons from solution. This may be attributed to the adsorption of protons from solution onto the PMN surface which should yield a positive surface charge at these low pH values. Such increase in the pH due to consumption of protons must be associated by the cation dissolution from the PMN surface at the acidic pH range. Such effect would be more pronounced with increasing solids content.

In the study mobility measurements were performed to investigate the change in dynamic mobility of PMN suspensions with time. Fig. 10 demonstrates the mobility curves for the suspension prepared at 0.1 vol% at pH 8 and 2. Measurements were repeated under the same conditions at 5 vol% to understand the effect of solids concentration on the dynamic mobility. It is clear from Fig. 10, the mobility of the particles increases with time in the case of the dilute suspension (0.1 vol%) at pH 2. On the other hand, the mobility of the particles in the case of 5 vol% is nearly constant through the measurement period. This behavior can be attributed to the higher dissolution rate of lead and magnesium ions in dilute suspension compared to the more concentrated one.

Previously Paik et al. reported that for barium titanate system, the dissolution process can be divided into three stages. First, incongruent dissolution in the first few minutes after initial contact with the acidic solution followed by a mass transport limited second step [15,17]. A similar dissolution process is valid for the PMN–water system. In the first step Pb$^{2+}$...
and Mg$^{2+}$ must diffuse out through a growing reactive surface layer. In the final stage, dissolution slows considerably and following possibilities may arise: a steady-state equilibrium concentration is reached, a thick depletion layer may form and leads to passivation; or the particles are completely converted to Nb-rich phase.

The FTIR spectra of the PMN powders aged in water at different conditions is shown in Fig. 11. Results suggest that the vibrational band over the range of 3200–3700 cm$^{-1}$ indicates the –OH groups in the sample. The band over the range of 500–900 cm$^{-1}$ corresponds to the metal oxygen bonds [18]. Wang et al. [8] analyzed the DRIFT spectra of the BaTiO$_3$ and PZT powders after aging in water for several days. They observed surface carbonate formation which is shown at 1450 and 1400 cm$^{-1}$ for barium titanate and lead zirconate titanate, respectively. Surface carbonate formation is an important problem faced during the aging of perovskites in water. It is well known that the presence of CO$_2$ naturally dissolved in the water contributes to BaTiO$_3$, instability and a tendency to form barium carbonate [19]. Nevertheless, in the current study surface carbonate formation was not observed for PMN powders.

It is a fact that local chemical inhomogeneities play an important role in abnormal grain growth in the perovskite structures exposed to the aqueous environment. Because broad distribution of the grain sizes may affect the desirable electric properties [11]. In our study it was shown that there is considerable amount of Pb$^{2+}$ and Mg$^{2+}$ ion dissolution from the PMN surface. Especially under very acidic conditions formation of niobium oxide-rich matrix was considered. Thus, there is a possibility for formation of A-site deficient structures if stoichiometric ratios were disturbed by ion dissolution. In order to investigate possible effects of ions dissolution from the oxide surface, PMN suspensions were stirred in water at room temperature at pH 9 and 6 and a subsequent heat treatment was applied.

![FTIR spectra of PMN powders aged in water](image1.png)

**Fig. 11.** FTIR spectra of PMN powders aged in water (a) no water treatment, (b) 24 h, pH 6, (c) 4 days pH 9 and (d) 24 h, pH 9.

![SEM micrographs of PMN powders aged in water followed by calcination at 1100 °C for 2 h](image2.png)

**Fig. 12.** SEM micrographs of PMN powders aged in water followed by calcination at 1100 °C for 2 h (A) No water treatment, (B) 4 days mixing in water, pH 9 and (C) 24 h mixing in water pH 6.
Fig. 12 denotes the SEM micrograph of the PMN powders mixed in water at different conditions followed by calcination at 1100 °C for 2 h (heating rate, 5 °C/min).

According to Fig. 12B due to low ion dissolution from oxide surface at pH 9 even very long contact times with water did not cause any detrimental effect in the microstructure of PMN. However calcination of the powders stirred in water at pH 6 for shorter period (24 h) caused a partial increase in the grain size (Fig. 12C).

Similarly, according to Fig. 13a, small amount of pyrochlore formation was observed in the sample aged in water at pH 6. On the other hand, X-ray diffraction spectra of the sample prepared at pH 9 showed that its crystal structure was pure perovskite and there is no formation of a second phase.

4. Conclusions

Chemical stability of lead magnesium niobate powders in water was investigated in this study. Solubility experiments showed that Pb$^{2+}$ and Mg$^{2+}$ ions dissolve from the PMN surface as a function of pH and solids concentration. Dissolution of Pb$^{2+}$ and Mg$^{2+}$ increased with decreasing pH value. Nevertheless, Nb$^{5+}$ dissolution was negligible under the same conditions. It was concluded that Pb–Mg–Nb–H$_2$O system undergoes incongruent dissolution with selective leaching and protective passive film forms on the solid surface.

FTIR spectra of the PMN powders aged in water at different conditions revealed that there is no surface carbonate formation. Microstructural characterization demonstrates ion dissolution from PMN surface at specified conditions did not cause any detrimental effect on the properties of the sintered product.

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